Using convolutional neural networks for stereological characterization of 3D hetero-aggregates based on synthetic STEM data

Lukas Fuchs¹, Tom Kirstein¹, Christoph Mahr², Orkun Furat¹, Valentin Baric^{3,4}, Andreas Rosenauer², Lutz Mädler^{3,4}, Volker Schmidt¹

¹Institute of Stochastics, Ulm University, 89069 Ulm, Germany

²Institute of Solid State Physics, University of Bremen, 28359 Bremen, Germany

 3 University of Bremen, Faculty of Production Engineering, 28359 Bremen, Germany

⁴Leibniz Institute for Materials Engineering IWT, 28359 Bremen, Germany

Abstract

The 3D nano/microstructure of materials can significantly influence their macroscopic properties. In order to enable a better understanding of such structure-property relationships, 3D microscopy techniques can be deployed, which are however often expensive in both time and costs. Often 2D imaging techniques are more accessible, yet they have the disadvantage that the 3D nano/microstructure of materials cannot be directly retrieved from such measurements. The motivation of this work is to overcome the issues of characterizing 3D structures from 2D measurements for hetero-aggregate materials. For this purpose, a method is presented that relies on machine learning combined with methods of spatial stochastic modeling for characterizing the 3D nano/microstructure of materials from 2D data. More precisely, a stochastic model is utilized for the generation of synthetic training data. This kind of training data has the advantage that time-consuming experiments for the synthesis of differently structured materials followed by their 3D imaging can be avoided. More precisely, a parametric stochastic 3D model is presented, from which a wide spectrum of virtual hetero-aggregates can be generated. Additionally, the virtual structures are passed to a physics-based simulation tool in order to generate virtual scanning transmission electron microscopy (STEM) images. The preset parameters of the 3D model together with the simulated STEM images serve as a database for the training of convolutional neural networks, which can be used to determine the parameters of the underlying 3D model and, consequently, to predict 3D structures of hetero-aggregates from 2D STEM images. Furthermore, an error analysis is performed with respect to structural descriptors, e.g., the hetero-coordination number. The proposed method is applied to image data of TiO₂-WO₃ hetero-aggregates, which are highly relevant in photocatalysis processes. However, the proposed method can be transferred to other types of aggregates and to different 2D microscopy techniques. Consequently, the method is relevant for industrial or laboratory setups in which product quality is to be quantified by means of inexpensive 2D image acquisition.

Keywords: synthetic HAADF-STEM, nanoparticle aggregate, hetero-aggregate, convolutional neural network, stereological characterization, stochastic 3D model, statistical image analysis

1 Introduction

The properties of many functional materials depend to a large extent on their structure and chemical composition. Hence, measuring both is mandatory in order to understand and improve their effective properties. An important class of materials are hetero-aggregates, which are compositions of at least two dissimilar classes of primary particles, called, for the sake of simplicity, particles from now on. Properties of hetero-aggregates can be quite different in comparison to aggregates that consist of monodisperse particles. A prominent example in applications concerned with photocatalysis are hetero-aggregates made of titanium dioxide (TiO₂) and tungsten trioxide (WO₃) [1, 2, 3]. The combination of both materials leads to aggregates with hetero-junctions, i.e., points at which two particles made from different materials touch. At such junctions, photogenerated electron-hole pairs are spatially separated, hindering their direct recombination, which results in a higher photocatalytic activity compared to pure TiO₂ [4].

In order to accurately investigate the properties of hetero-aggregates with imaging techniques, it is essential to resolve the individual particles within the structure. A suitable tool for the characterization of hetero-aggregates, consisting of particles with radii of a few nanometers, is (scanning) transmission electron microscopy, (S)TEM. With a spatial resolution in the sub-nanometer regime, even the atomic structure can be investigated. However, in conventional STEM only two-dimensional (2D) projection images of the aggregates can be acquired while information about the third dimension is lost. This problem can be overcome using STEM tomography, where the sample is tilted with respect to the electron beam such that a series of projection images under various projection angles is acquired, see [5]. From this series of STEM projection images, the three-dimensional structure can be reconstructed, e.g., with iterative reconstruction techniques [6]. The major disadvantage of STEM tomography is the fact that acquisition of a single tilt series can take several hours, and thus, this method does hardly allow for the investigation of a large number of aggregates. Furthermore, many samples do not allow for such a long measurement as hetero-aggregates and nanoparticles can change their structure and arrangement during extensive exposure to the electron beam, hindering the reconstruction.

As opposed to STEM tomography, 2D STEM images can be acquired within a few seconds, allowing for the acquisition of several images of various aggregates in a reasonable amount of time. For this reason, it is desirable to use 2D STEM images in order to characterize the 3D morphology of aggregates, such as their hetero-contacts [7] or their fractal dimension [8]. This can be achieved by training neural networks to predict structural properties of 3D hetero-aggregates from 2D STEM images. However, the training of neural networks requires a broad database of pairs of differently structured hetero-aggregate and corresponding 2D STEM images. The experimental acquisition of such a database, i.e., the synthesis of differently structured aggregates and their imaging would be expensive in both time and resources. Alternatively, simulated image data can be used for training purposes, see [9, 10] for a similar approach.

In the present paper, a stochastic 3D model for the generation of virtual aggregates and a physics-based STEM model for the simulation of corresponding 2D STEM images is combined in order to provide training data. In other words, methods of stochastic-geometry [11] are utilized to derive a parametric model for the generation of a wide spectrum of virtual, but realistic $\text{TiO}_2\text{-}\text{WO}_3$ hetero-aggregates. Note that the presented method is not limited to hetero-aggregates comprised of TiO_2 and WO₃. Particularly, details on the possibility to transfer the method to further materials are discussed in Section 5. Additionally, the virtual structures are passed to a physics-based simulation tool in order to generate virtual scanning transmission electron microscopy (STEM) images. The preset parameters of the 3D model together with the simulated STEM images serve as a database for the training of convolutional neural networks, which can be used to predict the parameters of the underlying 3D model and, consequently, to predict 3D structures of hetero-aggregates from 2D STEM images. In literature, there are already CNN-based approaches that do not use stochastic-geometry models to generate stochastically equivalent 3D structures from 2D images [12]. However, the presented approach aims to generate such digital shadows by combining a well-established parametric stochastic 3D model and a CNN-based approach. In order to use such a parametric stochastic stochastic stochastic stochastic stochastic stochastic stochastic stochastic approaches approaches that be presented approach approach. In order to use such a parametric stochastic stochastic stochastic stochastic stochastic stochastic approaches approaches the present paper is on this calibration procedure, also called model fitting.

More specifically, it is investigated how convolutional neural networks (CNNs) [13, 14] can be used to determine the parameters of the stochastic 3D model and, consequently, to generate digital shadows of 3D aggregates, from 2D STEM images. CNNs are a type of artificial neural networks commonly used in image analysis and recognition tasks, see e.g. [15]. They consist of multiple layers of neurons that learn to recognize patterns and features in the input data through a calibration process, called training. In the literature, several network architectures are noted for their effectiveness in image analysis tasks, including VGG [16, 17], Inception-v3 [18], and ResNet [19]. In particular, VGG is recognized for its deep but straightforward and computationally efficient architecture. Inception-v3 is known for its ability to extract multiscale features through its inception modules. ResNet addresses the vanishing gradient problem with its innovative use of residual connections [20].

In conventional spatial stochastic modeling of complex 3D morphologies, the process of model fitting typically involves several steps, see for example [21, 22]. First, image data has to be acquired, preprocessed, and segmented. Subsequently, an appropriate model type is chosen, and its model parameters are adjusted accordingly using descriptive statistics of the segmented image data. However, the approach considered in the present paper differs from the classical one. On the one hand, the image data does not have to be segmented, which is advantageous since image segmentation can be a time-consuming complex task. Moreover, the model parameters are predicted by the neural networks directly, meaning that the descriptive statistics are not chosen by hand. This allows for the use of stochastic 3D models with parameters that are not easily predictable from the image data.

In order to evaluate the performance of such a CNN-based approach, structural descriptors of aggregates drawn from the stochastic 3D model with preset parameter values are compared with structural descriptors of aggregates drawn from the 3D model with parameter values predicted by the CNN-based approach.

However, the structural similarity of the measured image data of aggregates and image data drawn from the fitted 3D model strongly depends on two factors, (i) the suitability of the chosen model type for the given data, and (ii) the ability of the selected CNN approach to determine the parameters of the stochastic 3D model from 2D STEM image data. More specifically, when analyzing measured image data of experimentally synthesized hetero-aggregates, there might not be any configuration of model parameters that results in a high-quality fit. In this case, the dissimilarities between the original image data and its digital shadows, generated by the fitted model, cannot necessarily be attributed to the fitting procedure, but rather to the inadequate choice of the model type. Thus, in the present paper, to be able to attribute these dissimilarities to an inadequate CNN approach, including data preprocessing, model architecture and learning procedure, the same stochastic 3D model is used as both the generator for the training data and the model to be fitted.

For an adequately designed CNN approach and adequately chosen type of the stochastic 3D model, digital shadows drawn from the fitted 3D model should be statistically equivalent to experimentally synthesized aggregates in terms of their 3D structure and chemical composition. Then, these digital shadows can be used as geometry input of (spatially resolved) numerical modeling and simulation, to determine their functional properties, see e.g. [23, 24]. Notably, there are recent attempts in literature that aim to reconstruct exact 3D structures from 2D data [25, 26], rather than statistically equivalent shadows. However, these methods are not able to generate a broad range of statistically similar 3D structures. In either way, 3D imaging techniques like STEM tomography of the aggregates can be avoided in order to derive quantitative process-structure or structure-property relationships for hetero-aggregates. Note that, by means of such relationships, optimized specifications of process parameters can be deduced, which lead to heteroaggregates with desired structures and properties. Digital shadows used for structure-property optimization are also referred to as digital twins. Their implementation will be the subject of a forthcoming study.

In summary, the present work aims to answer the following key research questions: Is it possible to predict the model parameters exclusively from 2D data, describing the 3D structure of hetero-aggregates? If so, how accurately can each of these parameters be predicted? Besides model parameters, how accurately can structural descriptors, e.g., coordination numbers, be predicted by the use of digital shadows?

The present work is organized as follows: In Section 2 the CNN-based approach is described to predict the 3D structure of hetero-aggregates from 2D STEM images. In particular, the generation of synthetic training data is explained, which are used for the prediction of model parameters. Then, in Section 3, we present results, which have

been obtained for various aspects of model parameter prediction. Section 4 compares the methods developed in the present paper with analysis tools considered in the literature. Section 5 concludes.

2 Methods

This section provides details how the presented CNN-based approach is built for predicting the 3D structure of hetero-aggregates from 2D STEM images. It comprises two main steps. First, virtual but realistic STEM images are generated from simulated 3D image data. More specifically, synthetic aggregates are drawn from a stochastic 3D model with preset model parameters, where the latter describe the aggregation procedure simulated by the model and therefore influence structural properties of the generated aggregates. These aggregates are then used to generate corresponding STEM images by means of a physics-based simulation tool, see Figure 1a. Systematically varying the parameters of the stochastic 3D model provides a wide range of differently structured aggregates and their STEM images. In a second step, visualized in Figure 1b, the parameters of the stochastic 3D model together with the simulated STEM images serve as a database for the training of CNNs, in order to learn how to reconstruct the parameters of the stochastic 3D model from STEM images. For the reconstruction, initially, a CNN extracts features from STEM images, which characterize the depicted structure of aggregates in an informative but not necessarily interpretable manner. Then, these features are utilized to predict our interpretable predefined model parameters. For more details, see Section 2.3. This approach is designed to allow for quick and accurate prediction of model parameters for real hetero-aggregates from measured STEM images and, consequently, to predict the 3D morphology of hetero-aggregates from 2D STEM images.

The quality of the predictor is evaluated with respect to the similarity between predefined and predicted model parameters. Recall that interpretable model parameters describe the aggregation procedure simulated by the stochastic model. Thus, a good match between predefined and predicted model parameters can already be an indication for a good structural match between aggregates generated by the model with predefined/predicted parameters. Nevertheless, some structural descriptors (i.e., quantities that characterize the structure of aggregates like hetero-coordination number) may be sensitive with respect to changes in the model parameters.

Therefore, the quality of the predictor is further evaluated by comparing structural descriptors of aggregates drawn from stochastic 3D models with pre-defined and predicted parameters, respectively, see Figure 1c,d. The structural descriptors considered in this paper, which are chosen due to their relevance in process engineering, are displayed in Table 1. They are complementary to the features, utilized in the model parameter prediction. Furthermore, these descriptors are interpretable and characterize the 3D structure of the aggregates (whereas the features describe the structure observed in 2D images).

descriptor	symbol
average cluster sizes of TiO_2 particles	$S_{\rm TiO_2}$
hetero-coordination number	$Z_{\rm hetero}$
coordination number	$Z_{\rm total}$

Table 1: Structural descriptors used for evaluating the model parameter prediction. For formal definitions of the descriptors, see Section 3.4.

2.1 Generation of synthetic training data

The use of synthetic training data requires careful attention to ensure that the artificially generated data accurately reflects particularities of experimentally measured data such that a regression model (e.g., a CNN) trained on synthetic data can be extended to new, real-world data. More precisely, if the generation of realistic data is successful, a network trained on this data can be used for applications on real-world data, and thus, reducing the amount of experimentally measured and labeled training data.

In the present study, synthetic training data was generated through a three-step process. First, virtual heteroaggregates were generated using a stochastic 3D model. Then, using a physics-based simulation tool, STEM intensities were determined based on the material and thickness of the aggregates. Finally, virtual but realistic STEM images were computed by adding noise and other sources of variability to the previously determined STEM intensities. In the following, the stochastic 3D model is introduced and then more details about each of the data generation steps mentioned above is provided.

training procedure



Figure 1: Workflow of the training and evaluation procedure.

2.1.1 Stochastic 3D model

In this section, we introduce the stochastic 3D model that will be used to generate a wide spectrum of virtual hetero-aggregates by varying the values of four different model parameters, denoted by $\theta_{D_f}, \theta_{\rho}, \theta_0, \theta_1$, where $\theta_{D_f} \in (1,3), \theta_{\rho} \in (0,1)$, and $\theta_0, \theta_1 \in \mathbb{N} = \{1, 2, \ldots\}$.

These model parameters control the fractal dimension, the mixing ratio, and clustering properties of the heteroaggregates, respectively.

Throughout this paper, a spherical particle is defined as a triplet p = (x, r, l) of particle position $x \in \mathbb{R}^3$, radius $r \in \mathbb{R}^+ = (0, \infty)$ and label $l \in \{0, 1\}$. Moreover, a hetero-aggregate A, consisting of N particles for some fixed $N \in \mathbb{N}$, is a set of connected and non-overlapping spherical particles, i.e.,

$$A = \{ p_i = (x_i, r_i, l_i) : x_i \in \mathbb{R}^3, \ r_i \in \mathbb{R}^+, \ l_i \in \{0, 1\}, \ 1 \le i \le N \}.$$

$$\tag{1}$$

In this context, two particles $p, p' \in A$ are said to be connected if for some $j \in \{2, ..., N\}$ there is a set of indices $\{i_1, ..., i_j\} \subset \{1, ..., N\}$ with $p = p_{i_1}$ and $p' = p_{i_j}$, such that

$$\|x_{i_k} - x_{i_{k+1}}\| \le 1.01(r_{i_k} + r_{i_{k+1}}) \text{ for all } k \in \{1, \dots, j\},$$
(2)

where $||y|| = \sqrt{\sum_{k=1}^{3} y_k^2}$ denotes the Euclidean norm of $y = (y_1, y_2, y_3) \in \mathbb{R}^3$. The prefactor 1.01 in Eq. (2) represents the maximum distance of particles that is allowed to consider them to be in contact. It is determined to be 1% of the sum of their radii. Moreover, two particles p = (x, r, l), p' = (x', r', l') are said to be overlapping if the distance of their centers is smaller than the sum of their radii, i.e., ||x - x'|| < r + r'. The label l of a particle p = (x, r, l) determines its material. More precisely, in our case, a particle with label l = 0 consists of WO₃, whereas a particle with label l = 1 consists of TiO₂.

The mixing ratio ρ of an aggregate A is defined as its fraction of particles with label l = 0, i.e.,

$$\rho(A) = \frac{\#\{p_i \in A \colon l_i = 0\}}{\#A},$$
(3)

where # denotes cardinality.

Notice the distinction in notation between θ_{ρ} and ρ since these values are not necessarily equal. More precisely, the model parameter θ_{ρ} can be set to an arbitrary value in the interval [0, 1] and it primarily influences the distribution of the structural descriptor ρ of aggregates generated with θ_{D_f} , as explained in more detail later on. Furthermore, the radius of gyration $R_g > 0$ of an aggregate A is given by

$$R_{\rm g} = \sqrt{\frac{\sum_{i=1}^{N} m_i \cdot \|x_i - c_0\|^2}{\sum_{i=1}^{N} m_i}}, \quad \text{with} \quad c_0 = \frac{\sum_{i=1}^{N} m_i x_i}{\sum_{i=1}^{N} m_i}, \tag{4}$$

where $m_1, \ldots, m_N > 0$ denote the particle masses and c_0 is the aggregate's center of mass.

The stochastic 3D model described below is motivated by the idea that hetero-aggregates have a fractal-like structure [27, 28]. This fractal-like structure of an aggregate A can be quantified by the so-called fractal dimension $D_{\rm f}$, given by

$$D_{\rm f} = \frac{\log\left(\frac{N}{k_{\rm f}}\right)}{\log\left(\frac{R_{\rm g}}{a}\right)},\tag{5}$$

where $k_{\rm f} > 0$ is a fractal prefactor, which is set to 1.3, and $a = \frac{1}{N} \sum_{i=1}^{N} r_i$ is the mean radius of the particles. For example, aggregates with a fractal dimension $D_{\rm f}$ close to 1 are arranged in a nearly straight line, whereas those with

a fractal dimension $D_{\rm f}$ close to 3 are composed of densely packed particles. Thus, realistic hetero-aggregates have values for $D_{\rm f}$ within the interval (1,3), see e.g. [27, 29, 30, 31, 32].

Note that the hetero-aggregate model presented in this paper is based on cluster-cluster aggregation, which involves a two-stage process for aggregate formation. In the first stage, primary particles aggregate to form small, homogeneous primary clusters. These primary clusters then undergo a second aggregation stage, leading to larger hetero-aggregates.

If an aggregate is homogeneous, i.e., all its particles share the same material, the labels $\{l_i\}_{i=1}^N$ will be neglected, and therefore the description of the aggregate A can be compressed to

$$A = \{ p_i = (x_i, r_i) : x_i \in \mathbb{R}^3, \ r_i \in \mathbb{R}^+, \ 1 \le i \le N \}.$$
(6)

In this case, the primary cluster model is introduced as a random set $\Phi_N = \{P_i : 1 \le i \le N\} \subset \mathbb{R}^3 \times \mathbb{R}^+$, which models the geometry of small homogeneous clusters of size N for some fixed $N \in \mathbb{N}$, compare [33] for earlier work. Here, $P_i = (X_i, R_i)$, where X_i is a random vector and R_i is a non-negative random variable describing the position and radius of a particle, respectively, for each $i \in \{1, \ldots, N\}$.

The random variables R_1, \ldots, R_N are independent and log-normally distributed with parameters $\mu = 12 \text{ nm}$ and $\sigma = 3 \text{ nm}$. However, the random vectors X_1, \ldots, X_N , which describe the particle positions, are recursively defined due to the dependency of X_i on X_1, \ldots, X_{i-1} and R_1, \ldots, R_i for all $1 < i \leq N$. This approach ensures that every realization of Φ_N is a set of connected and non-overlapping particles, with a predetermined fractal dimension D_f . Note that for technical reasons the random vector X_i can take not only values from \mathbb{R}^3 , but also the fictitious value ∞ . The latter value is used to model invalid particle positions.

More precisely, $X_1 = (0, 0, 0)$ and, under the condition that the values x_1, \ldots, x_i and r_1, \ldots, r_{i+1} of X_1, \ldots, X_i and R_1, \ldots, R_{i+1} are given for some $i \in \{1, \ldots, N-1\}$, the random vector X_{i+1} is uniformly distributed on some set $L(A, r_{i+1}) \subset \mathbb{R}^3$, provided that $(\infty, r) \notin A$ for all $r \in \mathbb{R}^+$ and $L(A, r_{i+1}) \neq \emptyset$, otherwise $X_{i+1} = \infty$. Here, $A = \{(x_1, r_1), \ldots, (x_i, r_i)\}$ and $L(A, r_{i+1}) \subset \mathbb{R}^3$ is the set of all permissible particle positions $x \in \mathbb{R}^3$ such that the set $A \cup \{(x, r_{i+1})\}$ describes a cluster of connected and non-overlapping particles with fractal dimension D_f being equal to some preset value $\theta_{D_f} \in (1, 3)$. In other words, $L(A, r_{i+1})$ is the set of positions where a particle of radius r_{i+1} can be added to the cluster A without violating the equation $D_f = \theta_{D_f}$. If no such position exists, X_{i+1} will be assigned ∞ , indicating that the cluster A cannot be extended.

To draw a sample from the random set $\Phi_N = \{P_i : 1 \leq i \leq N\} \subset \mathbb{R}^3 \times \mathbb{R}^+$, the procedure described above is repeated until $X_i \neq \infty$ for all i = 1, ..., N. The primary clusters generated in this way then undergo a second aggregation stage, leading to larger hetero-aggregates, which consist of N' primary clusters for some integer $N' \in \mathbb{N}$.

More formally, for some sequence of primary cluster sizes $N_1, \ldots, N_{N'}, N'$ independent random sets $\Phi_{N_1}^{(1)}, \ldots, \Phi_{N_{N'}}^{(n)}$ are considered as described above. The cluster $\Phi_{N_k}^{(k)}$ is assigned a random position C_k in $\mathbb{R}^3 \cup \{\infty\}$ for each $k \in \{1, \ldots, N'\}$, ensuring that realizations of the resulting hetero-aggregates are union sets of connected and nonoverlapping spheres, which adhere to a preset fractal dimension $D_f = \theta_{D_f}$. In the following, the cluster $\Phi_{N_k}^{(k)}$, which has been shifted by a (random) displacement vector C_k is denoted by $\Phi_{N_k}^{(k)} + C_k = \{(X + C_k, R) : (X, R) \in \Phi_{N_k}^{(k)}\}$. Furthermore, for each $k \in \{1, \ldots, N'\}$, the cluster $\Phi_{N_k}^{(k)} + C_k$ is assigned a (random) label L_k , which can be equal to 0 or 1, determining whether the cluster consists of WO₃ or TiO₂. The clusters of label 0 have a size of θ_0 whereas the clusters of label 1 have a size of θ_1 . These cluster sizes $\theta_0, \theta_1 \in \{1, \ldots, 6\}$ are a further model parameter. The labeled version of $\Phi_{N_k}^{(k)} + C_k$ with the (random) label L_k is denoted by $(\Phi_{N_k}^{(k)} + C_k) \times L_k = \{(X + C_k, R, L_k) : (X, R) \in \Phi_{N_k}^{(k)}\}$. Finally, the stochastic 3D model $\Psi_{N'}$ of hetero-aggregates, which consist of N' primary clusters, is given by

$$\Psi_{N'} = \bigcup_{k=1}^{N'} (\Phi_{N_k}^{(k)} + C_k) \times L_k.$$
(7)

Here, the random variables $L_1, \ldots, L_{N'}$, modeling the labels of the primary clusters, are independent and Bernoullidistributed with $\mathbb{P}(L_k = 1) = \frac{(1-\theta_p)\theta_0}{(1-\theta_p)\theta_0 + \theta_p \theta_1}$ for each $k \in \{1, \ldots, N'\}$. Note that the label of a primary cluster does not only determine its material but also its size. Specifically, the size N_k of the k-th primary cluster is given by $N_k = \theta_0 + L_k(\theta_1 - \theta_0)$ for each $k \in \{1, \ldots, N'\}$, i.e., a cluster has a size of θ_0 if its label is equal to 0, and θ_1 otherwise. For sufficiently large $N' \in \mathbb{N}$, according to the law of large numbers, these definitions of $L_1, \ldots, L_{N'}$ and $N_1, \ldots, N_{N'}$ ensure that the mixing ratios ρ of hetero-aggregates drawn from the stochastic 3D model $\Psi_{N'}$ are approximately equal to the preset value θ_{ρ} .

The random displacement vectors $C_1, \ldots, C_{N'}$ that describe the positions of primary clusters in the heteroaggregate model $\Psi_{N'}$ are again defined recursively to ensure that the particles of the random hetero-aggregate are connected and non-overlapping and that the fractal dimension θ_{D_f} is maintained. More precisely, C_1 is put to (0,0,0)and, given that $\bigcup_{k=1}^{i} (\Phi_{N_k}^{(k)} + C_k) = A_1$ and $\Phi_{N_i}^{(i+1)} = A_2$ for some $i \in \{1, \ldots, N'-1\}$, the random vector C_{i+1} is uniformly distributed on some set $\widetilde{L}(A_1, A_2) \subset \mathbb{R}^3$, provided that $(\infty, r) \notin A_1 \cup A_2$ for all $r \in \mathbb{R}^+$ and $\widetilde{L}(A_1, A_2) \neq \emptyset$, otherwise $C_{i+1} = \infty$. In this context, $\widetilde{L}(A_1, A_2) \subset \mathbb{R}^3$ is the set of all cluster positions $c \in \mathbb{R}^3$ for which the set $A_1 \cup (A_2 + c)$ represents a hetero-aggregate of connected and non-overlapping particles with fractal dimension D_f being equal to the preset value $\theta_{D_f} \in (1,3)$.

The resulting hetero-aggregate model $\Psi_{N'}$, which is described by the model parameters θ_{D_f} , θ_{ρ} , θ_{0} , θ_{1} can now be used to generate virtual aggregates. These aggregates consist of N' primary clusters with a fractal dimension of θ_{D_f} , an expected mixing ratio θ_{ρ} , and have a label-dependent clustering properties mainly influenced by θ_{0} and θ_{1} . Moreover, the model parameters have a multivariate influence on further structural descriptors, e.g. the heterocoordination number, see Section 3.4. In theory, this can be achieved by drawing samples from $\Psi_{N'}$ under the condition that $(\infty, r) \notin \Psi_{N'}$ for all $r \in \mathbb{R}^+$. However, due to computational limitations, this procedure can only be performed in an approximate sense. In the following section, this will be explained in detail.

2.1.2 Generation of virtual hetero-aggregates

The recursively defined models of primary clusters and hetero-aggregates described above can be used to construct algorithms for drawing samples from these models. More precisely, the simulation starts by selecting an initial particle (or cluster), to which particles (or clusters) are added sequentially. Each additional particle (or cluster) is assigned a random radius (or label) and placed at a uniformly sampled random position in L (or \tilde{L}), to be added to the existing cluster (or aggregate). This procedure is iterated until a desired cluster (or aggregate) size is reached. In the following, the desired size of each aggregate is independent, uniformly selected from the range $\{20, \ldots, 80\}$.

The sets $L, \tilde{L} \subset \mathbb{R}^3$ in the stochastic 3D model, from which particle (or cluster) positions are uniformly sampled in order to generate aggregates with a given fractal dimension, are only implicitly defined. Therefor, uniform sampling on L and \tilde{L} is computationally expensive.

To enable efficient uniform sampling from both $L(A, r_{i+1})$ and $\widetilde{L}(A_1, A_2)$, the radii of the particles in the sets $A \cup \{p_{i+1}\}$ and $A_1 \cup A_2$ are temporarily replaced by their respective arithmetic mean. Note that this replacement is used exclusively when calculating the fractal dimension D_f within the definitions of L and \widetilde{L} . Thus, all permissible positions for the center of mass of the added particle (or cluster) are located on the surface of a sphere around the center of mass of the cluster (or aggregate), the radius d of which is given by

$$d = \sqrt{\frac{a^2 (N_{\rm A} + N_{\rm C})^2}{N_{\rm A} N_{\rm C}}} \left(\frac{N_{\rm A} + N_{\rm C}}{k_f}\right)^{\frac{2}{D_{\rm f}}} - \frac{(N_{\rm A} + N_{\rm C})}{N_{\rm C}} R_A^2 - \frac{(N_{\rm A} + N_{\rm C})}{N_{\rm A}} R_C^2,\tag{8}$$

where N_A and N_C denote the number of particles in the aggregate and the cluster to be added, respectively, R_A and R_C are their respective radii of gyration, introduced in 4, and a and k_f are the quantities used in the definition of D_f given in Eq. (5), see also [32]. Since uniform sampling on the sphere surface can be performed efficiently, by means of rejection sampling, uniform sampling from the modified sets L or \tilde{L} can be done much faster. This procedure results in aggregates with fractal dimensions randomly distributed around the target value θ_{D_f} . For further details on the distribution of the fractal dimension $D_f(A)$ of an aggregate A generated by this model, see Section 2.2.1.

data acquisition the four model parameters θ_{D_f} , θ_{ρ} , θ_0 , θ_1 of the stochastic hetero-aggregate model are systematically varied, by name, the parameters regarding the fractal dimension θ_{D_f} , the intended mixing ratio θ_{ρ} and the primary cluster sizes θ_0 and θ_1 of the two materials. In this manner a broad spectrum of aggregates is obtained, which differ not only in preset model parameters used for their generation but also in structural descriptors like the ones listed in Table 1. The fractal dimension of TiO₂-WO₃ hetero-aggregates, which form by diffusion-limited cluster-clusteraggregation, is expected to approach the value of $D_f = 1.5$ for particles with dispersed sizes [30], and $D_f = 1.78$ for monodispersed particles [34]. Furthermore, the fractal dimension is expected to increase, when particles start to sinter at their contact points [31]. In order to create a large database of differently structured virtual heteroaggregates and their corresponding STEM images, the model parameter θ_{D_f} was varied in the present work from $\theta_{D_f} = 1.5$ to 2.5 in steps of 0.1. The intended mixing ratio θ_{ρ} was varied from $\theta_{\rho} = 0.1$ to 0.9 in steps of 0.1 and the primary cluster sizes θ_0 and θ_1 were chosen between one and six in steps of one for both materials, see also Table 2. Some examples of virtual hetero-aggregates for various values of the model parameters θ_{D_f} , θ_{ρ} , θ_0 , θ_1 are visualized in Figure 2.

Table 2: Range of model parameters. The model parameters θ_{D_f} and θ_{ρ} affect the fractal dimension and mixing ratio of the resulting aggregates, while the model parameters θ_0 and θ_1 determine the clustering behavior of the materials within the aggregates, respectively.



Figure 2: Examples of virtual hetero-aggregates. The labels correspond to the values of the vector $\theta = (\theta_{D_f}, \theta_{\rho}, \theta_0, \theta_1)$ of their model parameters. The WO₃ particles are displayed in blue, while the TiO₂ particles are displayed in orange. The particle sizes are drawn from a log-normal distribution with parameters μ and σ as defined above. Some primary clusters, as defined in Section 2.1.1, are highlighted.

2.1.3 Simulation of STEM intensities

After generating virtual hetero-aggregates, reference simulations to calculate the high-angle annular darkfield (HAADF)-STEM intensity of TiO₂ and WO₃ are conducted as a function of the sample thickness and material density. For that purpose, multi-slice simulations in the frozen-lattice approach [35] with the STEMSIM software [36] were performed. Simulations were done for the rutile and anatase phases of TiO₂ as well as for gamma and delta phases of WO₃. Crystal parameters and Debye-Waller factors were taken from [37, 38, 39, 40], and elastic atomic scattering amplitudes from [41] were used. The HAADF-STEM intensity for microscope parameters equal to those one would use in experiments with a ThermoFisher 60/300 Spectra microscope were simulated. This machine is equipped with a Cs-corrector for the probe forming system, an X-FEG and SuperXG2 EDXS detectors. A semi-convergence angle of $\beta = 21.1$ mrad and an acceleration voltage of 300 kV were set. The simulated HAADF-STEM intensity was obtained by integration of electrons scattered into the annular range between 55 mrad and 250 mrad after application of a detector specific sensitivity curve [36].

The HAADF-STEM intensity further depends on the orientation of the crystal with respect to the electron beam. To account for this effect, various orientations for each material and phase of the crystal were simulated.

Therefore, the crystal was systematically tilted in nine equal steps from a [100]- towards a [010]-viewing direction. In addition, a random tilt was simulated. The final result is a data set with the HAADF-STEM intensity as a function of the sample thickness for TiO_2 and WO_3 , each in two different crystal phases, each with ten orientations of the crystal with respect to the electron beam.

2.1.4 Generation of realistic STEM images

The third step combines the HAADF-STEM reference simulations described above with the virtual 3D heteroaggregates. STEM images show 2D projections of the aggregates, see Figure 3. Therefore, the projections of the individual particles along one direction are computed, as usual in electron microscopy, the electron beam direction and hence the projection direction is referred to as z-direction. This results in thickness maps for the individual particles. Using the reference simulations, these thickness maps are translated into maps of the HAADF-STEM intensities. To this end, for each particle, the reference simulation of the respective material was chosen in a random phase and a random orientation of the crystal with respect to the electron beam.

In an aggregate, which extends several tens of nanometers in z-direction, not all particles appear in focus. Only particles with centers located at height z = 0 nm are in focus, as the electron beam is focused on this plane. To account for this effect, each HAADF-STEM map of the individual particles is convolved with a Gaussian kernel. More precisely, for a particle located at height z, the standard deviation σ_{STEM} of the Gaussian kernel with which the corresponding HAADF-STEM map is convoluted is chosen as $\sigma_{\text{STEM}} = |z| \cdot \tan(\beta)$, where $\beta = 21.1$ mrad is the semi-convergence angle, assuming a conical beam shape. Then, blurred HAADF-STEM maps of individual particles are summed up to obtain the artificial HAADF-STEM image of the hetero-aggregate. Finally, shot noise according to a typical electron dose of 149 electrons/Å² [42] and scan noise according to a possible typical beam displacement of 0.01 nm [43] were applied.



Figure 3: Schematic representation of the 3D structure of a virtual hetero-aggregate (left) and its respective STEM image (right). The WO₃ particles are colored blue and correspond to the bright particles in the STEM image. The TiO_2 particles are colored orange and correspond to the dark particles.

2.2 Statistical analysis and processing of simulated data

In this section the need for the usage of neural networks is explained, addressing some problems connected with the reconstruction of preset values of the model parameters θ_{D_f} , θ_{ρ} , θ_0 , θ_1 , based on virtual aggregates drawn from the stochastic 3D model. Further complications in this reconstruction task arise when using 2D STEM data instead of the full 3D geometry of the aggregates, through a loss of information. Therefore, it is explained how image processing methods can be used to simplify the extraction of information from STEM data.

2.2.1 Estimating the parameters of the stochastic 3D model

One of the challenges associated with predicting the parameters of the stochastic 3D model from STEM images is that some model parameters are even imperceivable from the 3D structure of a virtual aggregate from which the corresponding STEM image is determined. This is due to the simplifying assumptions made within the simulation process of the 3D model and its stochastic nature, see Section 2.1.2, resulting in empirical values of the model parameters slightly differing from the preset ones.

For example, the fractal dimension $D_{\rm f}$ computed by means of Eq. (5) for a virtual hetero-aggregate A might differ from the preset value of the model parameter $\theta_{\rm D_f}$. More precisely, in the simulation of hetero-aggregates, the radii r_1, \ldots, r_N of particles considered in Eq. (5) are replaced by their arithmetic mean $(r_1 + \ldots + r_N)/N$, see Section 2.1.2. Also, the mixing ratio ρ of an aggregate A computed by means of Eq. (3) can deviate from the model parameter θ_{ρ} , due to the randomly chosen labels of primary clusters, modeled by the Bernoulli-distributed random variables $L_1, \ldots, L_{N'}$. For example, the first aggregate in Figure 2 has an expected (preset) mixing ratio of $\theta_{\rho} = 0.3$, but the actual mixing ratio ρ computed from Eq. (3) is $\rho = \frac{9}{41} \approx 0.22$. Recall that, in order to distinguish between these quantities, the vector of model parameters used to generate A is referred to as $\theta = (\theta_{\rm D_f}, \theta_{\rho}, \theta_0, \theta_1)$, while $D_{\rm f}(A)$ and $\rho(A)$ describe the empirical fractal dimension and mixing ratio of the aggregate A, computed from Eqs. (5) and (3), respectively.

Figure 4 illustrates the discrepancy between preset model parameters and the empirical fractal dimension and mixing ratio of virtual aggregates, computed from Eqs. (5) and (3), respectively. Nevertheless, Figure 4 indicates that, on average, the structural descriptors $D_{\rm f}$ and ρ nicely coincide with the preset model parameters $\theta_{\rm D_f}$ and θ_{ρ} . Therefore, rather than attempting to determine the model parameters $\theta_{\rm D_f}$, θ_{ρ} , θ_0 , θ_1 from a single aggregate A, a family $B = \{A_1, \ldots, A_{\nu}\}$ of $\nu > 1$ aggregates is used instead, called *batch* in the following. More specifically, it is expected that choosing a larger batch size would yield more accurate results, but at an increased cost.

We were not able to find any scalar features that can be utilized to predict the model parameters θ_0 and θ_1 associated with the cluster size used in the generation of virtual aggregates, i.e., in the cluster-cluster-model introduced in Section 2.1.1. For example, in order to predict the model parameter θ_0 , an obvious choice for such a scalar feature would be to describe the average size of observable WO₃ clusters, where an observable cluster is an inclusion maximal homogeneous subset $C \subset A$ of an aggregate A, i.e., there is no larger homogeneous subset $C' \subset A$ such that $C \subset C'$. These clusters can differ from the primary clusters used in the construction algorithm described in Section 2.1.1. Specifically, the observable clusters are formed by unions of primary clusters, whereas, contrary to the latter ones, the observable clusters are recognizable in the 3D data, see Figure 2 for a visualization. However, this average (observable) cluster size can not be used to predict θ_0 . Figure 5 shows that there are various specifications of model parameters that differ in θ_0 and, nevertheless, yield similar average cluster sizes of WO₃ particles.

The prediction of the model parameter vector θ is further complicated by the fact that only the 2D STEM image can be utilized, which may not perfectly inform the 3D morphology of A. To predict the preset vector of model parameters θ from a family B of aggregates using only their simulated STEM images, CNNs are initially utilized to extract relevant features from these images. These features are subsequently utilized to predict the preset model parameters, see the schematic description of this workflow shown in Figure 1b. While the process of extracting features from the STEM images remains largely consistent across all model parameters, the calculation of the estimators for θ_{D_f} , θ_{ρ} , θ_0 , θ_1 exhibits significant variations, see Sections 2.3.2-2.3.5 below. For instance, when estimating the model parameters θ_{D_f} and θ_{ρ} , the features computed from a STEM image I of an aggregate A are scalar values that approximate $D_{\rm f}(A)$ and $\rho(A)$. Then, the arithmetic mean of the respective image-wise features of a family $B = \{A_1, \ldots, A_\nu\}$ of aggregates is used as estimators $\hat{\theta}_{\rm D_f}$ and $\hat{\theta}_{\rho}$ for $\theta_{\rm D_f}$ and θ_{ρ} . In contrast, when predicting the model parameters θ_0 and θ_1 , a neural network is employed to identify high-dimensional features from which the estimators $\hat{\theta}_0$ and $\hat{\theta}_1$ for θ_0 and θ_1 are computed, see Section 2.3 below.



Figure 4: Visualization of empirical probability densities of the fractal dimension $D_{\rm f}(A)$ (left) and mixing ratio $\rho(A)$ (right) of virtual hetero-aggregates, depending on the model parameters $\theta_{\rm D_f}$ and θ_{ρ} , where the values of $D_{\rm f}(A)$ and $\rho(A)$ are computed by means of Eqs. (5) and (3), respectively. The other model parameters were chosen at random from their respective ranges, as introduced in Section 2.1.2. To improve clarity, kernel density estimation was used to assign colors to the scatter points computed for 19 440 realizations of the stochastic 3D model.



Figure 5: Average cluster size of WO₃ particles. Each scatter point is computed over a batch $B = \{A_1, \ldots, A_\nu\}$ with $\nu = 12$ for six different specifications of the model parameters $\theta_{D_f}, \theta_{\rho}, \theta_0, \theta_1$. For three of these specifications, the values of $\theta = (\theta_{D_f}, \theta_{\rho}, \theta_0, \theta_1)$ and the average cluster sizes, corresponding to the red scatter points, are displayed.

2.2.2 Data processing and augmentation

Various common image processing methods are used to simplify the extraction of information from STEM images. In particular, the pixel intensity values of STEM images are linearly scaled to the entire range of [-0.5, 0.5] and rounded to 256 equidistant values in order to achieve faster convergence to a lower error during the training process. More specifically, the scaling centers the pixel intensity values around zero [44], whereas the rounding reduces the noise of the images. Note that this procedure is performed on all STEM images, even if not explicitly mentioned, whereas the subsequent preprocessing steps will only be applied during training.

Overfitting is a common problem where neural networks achieve good results on training data but perform rather poorly when applied to previously unseen data. This can occur when the model learns irrelevant information within the dataset. As a result, the model fits too closely to the training set and becomes overfitted, making it unable to generalize well to new data. To address this issue, augmentation of training data is used. In the context of the present paper, this means that the input data is randomly modified in each training step, such that during each step of the training procedure the network is provided with input data, which differs from the input data of previous steps. Therefore, a significantly larger number of training steps can be conducted while still providing the neural network with novel training data in each step, and thus, avoiding overfitting.

Note that there is a wide variety of possible methods for modifying input data, which are commonly used in training data augmentation, e.g., rotation, reflection, radial transformation, elastic distortion [45] and random erasing [46]. However, in order to preserve certain structural descriptors of aggregates observed in image data, like shape and size descriptors of particles, only random rotations, reflections and small displacements are used for training data augmentation.

2.3 CNN-based approach for the prediction of model parameters

The goal of this section is to introduce the CNN-based methodology for predicting the model parameters θ_{D_f} , θ_{ρ} , θ_0 , θ_1 of the hetero-aggregate model from (simulated) STEM images.

Due to computational constraints, it was not feasible to generate the required number of aggregates for each possible preset of the model parameters $\theta_{D_f}, \theta_{\rho}, \theta_0, \theta_1$. Therefore, in order to ensure robust training, the focused was on generating 100 aggregates for each triple $(\theta_{\rho}, \theta_0, \theta_1)$ in $\{0.1, \ldots, 0.9\} \times \{1, \ldots, 6\} \times \{1, \ldots, 6\}$, as these parameters exhibited interactive effects that were crucial for our study. More specifically, for each such triple, two values $\theta_{D_f}^{(1)}, \theta_{D_f}^{(2)}$ of θ_{D_f} were chosen at random from $\{1.5, \ldots, 2.5\}$ and each resulting model parameter preset $(\theta_{D_f}^{(1)}, \theta_{\rho}, \theta_0, \theta_1), (\theta_{D_f}^{(2)}, \theta_{\rho}, \theta_0, \theta_1)$ was used to generate 50 aggregates. After applying the STEM simulation described in Section 2.1, this results in a set $G = \{(A_i, I_i, \theta_i) : 1 \le i \le 32400\}$ of 32400 triplets of 3D aggregates A_i , corresponding STEM images I_i , and vectors of preset model parameters $\theta_i = (\theta_{D_f,i}, \theta_{\rho,i}, \theta_{0,i}, \theta_{1,i})$. The set G is thereafter split into two datasets, one for training and one for evaluation.

For both, training and evaluation, batches

$$B = \{ (A_{i_1}, I_{i_1}, \theta_{i_1}), \dots, (A_{i_n}, I_{i_n}, \theta_{i_n}) \} \subset G$$
(9)

will be used for some $\nu > 1$, which are generated by the same preset of model parameters, i.e., $\theta_{i_1} = \ldots = \theta_{i_\nu}$. To ensure the availability of such batches, the split of G is done such that there is no model parameter configuration that occurs less than 20 times in neither the data used for training nor in the data used for evaluation. These two datasets will be referred to by their respective index sets T (for training) and E (for evaluation), where $T \cup E = \{1, \ldots, 32400\}$ with #T = 19440 and #E = 12960.

In the following, it is explained how the triplets (A_i, I_i, θ_i) are used to generate pairs of image data and ground truth labels, which will be utilized for the training of the neural networks. First, general aspects of network architecture and training are presented and, then, some specifics regarding the prediction of each of the four model parameters $\theta_{D_f}, \theta_{\rho}, \theta_0, \theta_1$ are given.

2.3.1 Network architecture and training

The networks used for feature extraction are all based on the same basic network architecture, which is a slight modification of the VGG architecture [16, 47], regardless of the model parameter being predicted. More precisely, it consists of stacked convolutional layers with a kernel size of 3×3 , batch normalization layers [48], the ReLu activation function, given by

$$\operatorname{ReLu}(x) = \max\{0, x\} \quad \text{for } x \in \mathbb{R},$$

$$(10)$$

and max pooling layers with a kernel size of 2×2 , followed by fully connected layers. The basic architecture of the convolutional neural networks considered in the following has the form

$$CNN = g(f), \tag{11}$$

i.e., it is represented as the composition of two subnetworks, f and g. The subnetwork f consists of the convolutional part of the basic network architecture, a flatten layer and two dense layers with a final output dimension of 112. The subnetwork g consists of two dense layers with a final output dimension of 1. A schematic representation of the network architecture is given in Figure 6 (left), whereas details regarding this architecture are provided in Table 3. Note that the modular structure of the proposed architecture allows for easy replacement of the subnetwork f with other network architectures, to address specific needs. These considerations include computational speed, tendencies to overfit on synthetic data, and other factors that influence the generalization to real data. For a quantitative comparison of the predictive power for a selection of some alternative network architectures [20], the reader is referred to Section 4.

To achieve a high prediction quality, the parameters of the neural networks have to be adopted. This will be done supervised. More precisely, the dissimilarity between the ground truth, denoted as $y = (y_1, \ldots, y_n)$, and the network output $\hat{y} = (\hat{y}_1, \ldots, \hat{y}_n)$, i.e., $\hat{y}_1 = \text{CNN}(x_1), \ldots, \hat{y}_n = \text{CNN}(x_n)$ for some input $x = (x_1, \ldots, x_n), n > 1$, will be minimized.

For example, when predicting the fractal dimension θ_{D_f} , the input x of the network consists of STEM images $I_1, ..., I_n$ and the ground truth is given by the vector of fractal dimensions of the respective aggregates $A_1, ..., A_n$, i.e. $y = (D_f(A_1), ..., D_f(A_n))$. The ground truth and the network output are compared using the mean squared error (MSE), given by

$$MSE(y, \hat{y}) = \frac{1}{n} \sum_{i=1}^{n} (y_i - \hat{y}_i)^2.$$
 (12)

The resulting loss $MSE(y, \hat{y})$ is minimized by a gradient descent method using an Adam optimizer [13, 49] with a learning rate of 0.0001, where the value of n in Eq. (12) determines the number of network evaluations before a step of the gradient descent method is applied. These evaluations are done on the training data, given by the index set T, where n is put to 16 when predicting θ_{D_f} or θ_{ρ} , and n = 8 otherwise.

The general network architecture described above and the prediction procedure will be slightly adapted for each of the four model parameters θ_{D_f} , θ_{ρ} , θ_0 and θ_1 . In the following, detailed explanations will be provided regarding these parameter-specific adaptations.



Figure 6: The basic network architecture represented as a simple composition of subnetworks f and g (left), and an adjusted multi-image input network (right), where the subnetwork f is applied to multiple input images and the outputs are concatenated as a new input for subnetwork g (see also Section 2.3.4 below).

	layer	output shape	number of parameters
0	input	$(512, 512, 1) \cdot b$	$0 \cdot 1$
1	average pooling	$(256, 256, 1) \cdot b$	$0 \cdot 1$
2	convolution + batch norm. + ReLu	$(254, 254, 8) \cdot b$	$(80+56+0)\cdot 1$
3	convolution + batch norm. + ReLu	$(252, 252, 16) \cdot b$	$(1168 + 112 + 0) \cdot 1$
4	convolution + batch norm. + ReLu	$(250, 250, 32) \cdot b$	$(4640 + 224 + 0) \cdot 1$
5	max pooling	$(125, 125, 32) \cdot b$	$0 \cdot 1$
6	convolution + batch norm. + ReLu	$(123, 123, 64) \cdot b$	$(18496 + 448 + 0) \cdot 1$
7	convolution + batch norm. + ReLu	$(121, 121, 64) \cdot b$	$(36928 + 448 + 0) \cdot 1$
8	max pooling	$(60, 60, 64) \cdot b$	$0 \cdot 1$
9	convolution + batch norm. + ReLu	$(58, 58, 128) \cdot b$	$(73856 + 896 + 0) \cdot 1$
10	convolution + batch norm. + ReLu	$(56, 56, 128) \cdot b$	$(147584 + 896 + 0) \cdot 1$
11	max pooling	$(28, 28, 128) \cdot b$	$0 \cdot 1$
12	convolution + batch norm. + ReLu	$(26, 26, 256) \cdot b$	$(295168 + 1792 + 0) \cdot 1$
13	convolution + batch norm. + ReLu	$(24, 24, 256) \cdot b$	$(590080 + 1792 + 0) \cdot 1$
14	max pooling	$(12, 12, 256) \cdot b$	$0 \cdot 1$
15	flatten	$(36864) \cdot b$	$0\cdot 1$
16	dense + ReLu	$(224) \cdot b$	$8257760\cdot 1$
17	dense + ReLu	$(112) \cdot b$	$25200\cdot 1$
18	dense + ReLu	(44)	$112 \cdot b \cdot 44 + 44$
19	dense	1	$45 \cdot 1$
			$= b \cdot 4928 + 9457713$

Table 3: Details of network architecture. The value of b is put equal to b = 1 for the prediction of the model parameters θ_{D_f} and θ_{ρ} , and $b = \nu$ for the prediction of θ_1 and θ_0 . Since padding is omitted, each convolutional layer reduces the size of the feature map by two in both dimensions.

2.3.2 Fractal dimension

The fractal dimensions $D_{\rm f}(A_{i_1}), \ldots, D_{\rm f}(A_{i_{\nu}})$ of the aggregates $A_{i_1}, \ldots, A_{i_{\nu}}$ in a batch B, as introduced in Eq. (9), are typically symmetrically distributed around the preset value of $\theta_{\rm D_f}$, which will be denoted by $\theta_{\rm D_f}(B)$ in the following, see Figure 4a. Therefore, the mean value $\overline{D}_{\rm f}(B)$, given by

$$\overline{D}_{\mathbf{f}}(B) = \frac{1}{\nu} \sum_{k=1}^{\nu} D_{\mathbf{f}}(A_{i_k}),$$

could be used as an estimator for $\theta_{D_f}(B)$. However, since the fractal dimensions $D_f(A_{i_1}), \ldots, D_f(A_{i_\nu})$ cannot be directly determined from the STEM images $I_{i_1}, \ldots, I_{i_\nu}$, approximations $\hat{D}_f(I_{i_1}), \ldots, \hat{D}_f(I_{i_\nu})$ are used instead. These approximations are computed by a convolutional neural network CNN^{D_f}, where the STEM images $I_{i_1}, \ldots, I_{i_\nu}$ are used as input. Thus, finally, the estimator $\hat{\theta}_{D_f}(B)$ for $\theta_{D_f}(B)$ is given by

$$\widehat{\theta}_{D_{f}}(B) = \frac{1}{\nu} \sum_{j=1}^{\nu} \widehat{D}_{f}(I_{i_{j}}) = \frac{1}{\nu} \sum_{j=1}^{\nu} \text{CNN}^{D_{f}}(I_{i_{j}}).$$
(13)

The architecture of the neural network $\text{CNN}^{D_{\text{f}}}$ coincides with the one described in Section 2.3.1. The activation function of the output layer is a scaled sigmoid function. This kind of activation function is a standard choice for NNs with bounded outputs. More precisely, the activation function is given by $\gamma(x) = \alpha \frac{1}{1+e^{-x}} + \beta$ for $x \in \mathbb{R}$, where $\alpha = 1.4$ and $\beta = 1.3$ are selected to ensure that the network can represent the expected range of values for D_{f} , with added tolerances on each side of the expected range, see Figure 4a. Note that the input of the network during training consists of augmented versions $a(I_i)$ of the STEM images I_i for $i \in T$, i.e., images that arise from I_i by reflecting, rotating and displacing, as described in Section 2.2.2. The corresponding supervisory signal consists of the fractal dimension of the corresponding aggregates. Hence, the network training is conducted using pairs $(a(I_i), D_{\text{f}}(A_i))$, for $i \in T$.

2.3.3 Mixing ratio

In Figure 4b the distribution of the mixing ratio of aggregates in dependence of the model parameter θ_{ρ} is visualized. From there, it is evident that the mixing ratios $\rho(A_{i_1}), \ldots, \rho(A_{i_{\nu}})$ of aggregates $A_{i_1}, \ldots, A_{i_{\nu}}$ within a batch B, generated by the 3D model with a preset value of $\theta_{\rho}(B)$, follow a distribution the mean of which is approximately equal to $\theta_{\rho}(B)$.

This suggests using a similar approach as described above in Section 2.3.2. However, note that there are some aggregates with a mixing ratio $\rho(A_i)$ being equal to 0 or 1. A neural network with an architecture as that of $\text{CNN}^{\text{D}r}$ does not reflect these discrete values properly. Therefore, the prediction procedure for the mixing ratio is slightly modified by initially classifying whether an image I_i depicts an aggregate with mixing ratio of exactly 0 or 1, using a classification network $\text{CNN}^{\rho}_{\text{class}}$, and afterwards predicting the mixing ratio of the corresponding aggregate, using a regression network $\text{CNN}^{\rho}_{\text{reg}}$. For this purpose, the networks $\text{CNN}^{\rho}_{\text{class}}$ and $\text{CNN}^{\rho}_{\text{reg}}$, having the same basic network architecture as described in Section 2.3.1 and a commonly used [50] unscaled sigmoid function $\gamma(x) = \frac{1}{1+e^{-x}}$ for $x \in \mathbb{R}$ as activation function in the output layer, are trained for the respective tasks. The training of the regression network $\text{CNN}^{\rho}_{\text{reg}}$ is done on pairs $(a(I_i), \rho(A_i)), i \in T$, of augmented STEM images and corresponding ground truth mixing ratios, while the training of the classification network $\text{CNN}^{\rho}_{\text{class}}$ is done on pairs of augmented STEM images and corresponding aggregate as heterogeneous or homogeneous, respectively.

However, it is a well-known problem that number-wise imbalanced classes can lead to poorly performing classifications since classifiers tend to neglect the underrepresented classes, also known as imbalance problem [51]. To address this issue, the augmented STEM images of homogeneous aggregates, which account for about 10% of all images, were oversampled in the training procedure of the classifier to achieve balanced classes.

Finally, to predict the mixing ratio of an aggregate via its STEM image, the outputs of $\text{CNN}_{\text{class}}^{\rho}$, which identifies homogenous aggregates, and $\text{CNN}_{\text{reg}}^{\rho}$, which determines the mixing ratio, are combined. More specifically, for a STEM image *I*, the predicted mixing ratio $\hat{\rho}(I)$ of the corresponding aggregate is given by

$$\widehat{\rho}(I) = \begin{cases} \eta \left(\text{CNN}_{\text{reg}}^{\rho}(I) \right), & \text{if } \text{CNN}_{\text{class}}^{\rho}(I) > 0.5, \\ \text{CNN}_{\text{reg}}^{\rho}(I), & \text{else,} \end{cases}$$

where $\eta : [0,1] \to \{0,1\}$ is the function that rounds a number $x \in [0,1]$ to its closest integer $\eta(x) \in \{0,1\}$. This results in the estimator $\hat{\theta}_{\rho}(B)$ for the preset model parameter $\theta_{\rho}(B)$ of a batch $B = \{(A_{i_1}, I_{i_1}, \theta_{i_1}), \dots, (A_{i_v}, I_{i_v}, \theta_{i_v})\}$, given

$$\widehat{\theta}_{\rho}(B) = \frac{1}{\nu} \sum_{j=1}^{\nu} \widehat{\rho}(I_{i_j}).$$
(14)

2.3.4 Size of primary WO₃ clusters

In the procedures for predicting the model parameters θ_{D_f} and θ_{ρ} , described above, the process of determining an estimator involved the identification of a scalar feature that describes an aggregate property, namely, the fractal dimension D_f and the mixing ratio ρ , which is predominantly influenced by the corresponding model parameter. This scalar feature can be directly computed from the virtual 3D aggregates, and thus, it is possible to predict it from the corresponding 2D STEM images. Consequently, using this scalar feature, formulas for estimating the model parameter from this property has been derived, see Eqs. (13) and (14).

Since the model parameter θ_0 is designed to control the cluster sizes of WO₃ particles for the cluster-clusteraggregation model introduced in Section 2.1.1, such a property should relate to the number of connected WO₃ particles. However, the sizes of observable clusters are not only influenced by θ_0 but also by θ_{ρ} . On the one hand, larger values of θ_0 lead to larger primary cluster sizes of clusters of label 0 and thus larger observable clusters. Lower values of θ_{ρ} lead to larger proportions of primary clusters of label 0. Therefore, it is more likely that two primary clusters that are in contact, share the material label 0, and thus, the expected size of observable clusters of label 0 increases, see Figure 5.

This makes the average of the observable cluster size on its own an unsuitable property for estimating the model parameter θ_0 . Therefore, one has to search for another feature that is functionally related to θ_0 . Additionally, a functional relationship that suitably maps features derived from STEM images to an estimator of θ_0 may not be captured solely by an average, necessitating the search for another suitable function. However, these two steps can be quite complex and time-consuming if done heuristically. To address this, a data-driven approach utilizing a neural network is adopted. This approach allows us to determine the feature vectors and the formula that relates them to the corresponding model parameter θ_0 . More specifically, the identification of relevant features is conducted by means of part f of the basic network architecture described in Section 2.3.1. The subnetwork f is applied to all images in a batch individually, and the concatenated results are then used as input of part g of the basic network architecture, which is in charge of determining the relationship between the feature vectors determined by f and the model parameter θ_0 . In detail, this results in a modified network, denoted as CNN⁰, which is given by

$$\operatorname{CNN}^{0}(I(B)) = g(f(I_{i_{1}}), \dots, f(I_{i_{\nu}})),$$
(15)

where $I(B) = \{I_{i_1}, \ldots, I_{i_\nu}\}$ denotes the STEM images corresponding to the aggregates $A_{i_1}, \ldots, A_{i_\nu}$ in a batch B. Referring to Table 3, the feature vectors of STEM images up to the output of layer 17 are computed as before. Then these feature vectors of a batch are concatenated and used as input of layer 18. The final output layer uses a ReLu transfer function. The modified network architecture is illustrated on the right-hand side of Figure 6.

Note that the approach described above differs from the commonly used technique where a network, denoted as f', takes multi-channel input data, i.e., in our case $f'(I_{i_1}, ..., I_{i_{\nu}})$. Such an approach allows the network to detect spatially resolved interdependencies among the images. In contrast, our approach considered in Eq. 15 employs identical CNNs f for dimensionality reduction and feature extraction on each input channel individually. As a consequence, this ensures uniform feature extraction for every input image while also reducing the number of trainable parameters in the CNN. The choice of this approach is rooted in the concept that each image within a batch a priori contains the same information regarding the underlying model parameters, and the lack of spatial interdependence between the images, which would be relevant for the prediction of model parameters.

Due to the problem-specific architecture of the network CNN^0 , the training data no longer consists of pairs of individual images and corresponding ground truths. Instead, for each batch B, the training pair $(\{a(I_{i_1}), \ldots, a(I_{i_{\nu}})\}, \theta_0(B))$ consists of a corresponding batch of augmented images and the underlying model parameter $\theta_0(B)$.

by

Since the model parameter θ_0 can only take integer values, the output of the network has to be rounded to obtain a valid estimator for θ_0 , given by $\hat{\theta}_0(B) = \eta(\text{CNN}^0(I(B)))$, where where $\eta : [0, \infty) \to \{0, 1, ...\}$ is the function that rounds a number $x \ge 0$ to its closest integer $\eta(x) \in \{0, 1, ...\}$.

2.3.5 Size of primary TiO₂ clusters

The method used to predict the model parameter θ_1 for the cluster size of TiO₂ particles is similar to the approach described in the previous section. Nonetheless, given that the pixel intensity values of TiO₂ particles in the STEM images closely resemble the background and are significantly lower than those of WO₃ particles, they are considerably more difficult to differentiate by visual inspection. Consequently, it might be plausible that a neural network could also encounter challenges in tasks that depend on the identification of TiO₂ particles. As shown in Figure 7, the neural network CNN¹ achieves unsatisfactory results when using unadjusted image data, which may be due to the difficulty mentioned above.

To address this issue, the intensity value p > 0 of non-background pixels in the STEM images is replaced by its multiplicative inverse p^{modified} , i.e., for some threshold t > 0 the modified pixel value is given by

$$p^{\text{modified}} = \begin{cases} p, & \text{if } 0$$

This procedure is applied to all STEM images used in the prediction of θ_1 before the preprocessing steps described in Section 2.2.2 are applied. The highlighting effect of this adjustment of pixel intensity values is shown in Figure 8.



Figure 7: Prediction error of the network CNN¹ for unmodified input images, in dependence of the preset value of the model parameter θ_1 . The prediction quality is comparable to that of a constant prediction, where CNN¹(I(B)) = 3.5. The results have been obtained on a randomly chosen subset of the training data set.



Figure 8: Effect of pixel intensity modification. The image on the right-hand side is obtained after replacing the intensity values of non-background pixels (shown on the left-hand side) by their multiplicative inverse, where a threshold of t = 0.001 is used.

3 Results

In this section, the results of the analysis on various aspects of model parameter prediction are presented. To ensure that these results accurately represent the generalization capability of the trained neural networks, all evaluations were conducted on data not used during training. More specifically, recall that the data corresponding to the index set T is used for training, whereas the data corresponding to the index set E is used to evaluate results, see Section 2.3 for details on the training-test split.

As a prelude to the main findings, first, the impact of batch size on prediction quality is assessed for all four model parameters θ_{D_f} , θ_{ρ} , θ_0 , θ_1 . For that purpose, Figure 9 illustrates how the batch size affects the quality of the predictions with respect to the mean absolute error (MAE), defined as

$$MAE(y,\hat{y}) = \frac{1}{n} \sum_{i=1}^{n} |y_i - \hat{y}_i|, \qquad (16)$$

where n > 0 is the number of predictions and $\hat{y} = (\hat{y}_i)_{i=1,...,n}$ are the predictions of the ground truth values $y = (y_i)_{i=1,...,n}$. Note that the mean absolute error given in Eq. (16) yields more easily interpretable values compared to the MSE, which was used for network training, see Section 2.3. As expected, it can be observed that larger batch sizes lead to better predictions. However, no significant improvement is observed for values exceeding 10. Thus, the results presented below, which were computed with a fixed batch size of $\nu = 12$, can be considered representative for the presented methodology.

3.1 Fractal dimension

The accuracy of the estimator $\hat{\theta}_{D_f}$ for θ_{D_f} depends on two key properties. First, the mean error of the single STEM image predictions $\text{CNN}^{D_f}(I)$ should be centered around zero, since otherwise a bias could be propagated through the averaging procedure and therefore bias the estimator $\hat{\theta}_{D_f}$, see Eq. (13). Second, the variance of the single image prediction error should be low, so a low variance estimator can be achieved even with a small batch size ν .

In Figure 10a the error for the predicted fractal dimension $D_{\rm f}$ is shown. As desired, the error of the network output exhibits a small absolute value for the bias and a low variance, as indicated by a mean value of -0.006 and interquartile range of 0.118. As the network output is a suitable basis for predicting the model parameter $\theta_{\rm D_f}$, the estimator $\hat{\theta}_{\rm D_f}$ achieves an MAE of 0.041, see Figure 10b. The network tends to slightly overestimate the fractal dimension of the depicted aggregates for small preset values of $\theta_{\rm D_f}$ and underestimate it for large ones. This behavior is further pronounced in the estimator $\hat{\theta}_{\rm D_f}$. For a possible explanation of this trend, see Section 4 below.



Figure 9: Quality of the estimators for θ_{D_f} , θ_{ρ} , θ_0 , θ_1 , in dependence of the batch size ν . Due to different orders of magnitude, the error curves for θ_{D_f} , θ_{ρ} and θ_0 , θ_1 are shown separately. The MAEs, defied by Eq. 16, are computed over all available evaluation data, indexed by the set E.



Figure 10: Prediction error of $\widehat{D}_{\rm f}$ (left) and $\widehat{\theta}_{\rm D_f}$ (right). In Subfigure (a) the quality of the prediction of the fractal dimension per aggregate is visualized, where the colors are computed by means of a Gaussian kernel density estimator. In Subfigure (b), the error regarding the prediction of the model parameter $\theta_{\rm D_f}$ is shown, where batches of size $\nu = 12$ are used for the computation of $\widehat{\theta}_{\rm D_f}$.

3.2 Mixing ratio

To evaluate the accuracy of the estimator for the model parameter θ_{ρ} , first, the image-wise straightforward case is considered, where the output $\text{CNN}_{\text{reg}}^{\rho}(I)$ of the regression network is used as an estimator for the mixing ratio $\rho(A)$, without considering the classification network. As shown in Figure 11a, the output $\text{CNN}_{\text{reg}}^{\rho}(I)$ of the regression network exhibits a relatively high bias for aggregates A such that $\rho(A) \in [0, 0.1]$ or $\rho(A) \in [0.9, 1]$, with biases of about 0.04 and -0.1, respectively.

To address this issue, in Section 2.3.3 a procedure, which utilizes an additional classification network $\text{CNN}_{\text{class}}^{\rho}$ is presented. In Figure 11b, the resulting image-wise error of $\hat{\rho}$ using this procedure is shown. It is evident that the error of $\hat{\rho}$ is significantly reduced for homogenous aggregates. More precisely, the bias of $\hat{\rho}(I)$ for aggregates A with $\rho(A) \in [0, 0.1]$ or $\rho(A) \in [0.9, 1]$ decreases to about 0.009 and 0.02, respectively. Incorporating the additional network, the MAE of the image-wise predicted mixing ratio $\hat{\rho}(I)$ of an aggregate A decreases from 0.059 to 0.053.

Consequently, the MAE of the batch-wise prediction $\hat{\theta}_{\rho}$ of θ_{ρ} improves significantly, reducing from 0.027 to 0.017.

Note that the diagonally arranged points in Figure 11b are due to a small number of falsely classified heterogeneous aggregates, whereas the significantly thinned vertical lines are due to correctly classified homogeneous aggregates. The amounts of correctly and falsely classified aggregates are displayed in Table 4.



Figure 11: Prediction error of $\hat{\rho}$ (left, center) and $\hat{\theta}_{\rho}$ (right). In Subfigure (a) the error of the unrounded output of the regression network is displayed per image. The black line, visualizing the mean error, is computed via a sliding window. The error for the output of the modified network is shown in Subfigure (b). In Subfigure (c) the error regarding the prediction of the model parameter θ_{ρ} is displayed, where the modified values of $\hat{\rho}$ are used.

prediction $\setminus 1$	true	homogenous	heterogeneous	
homogenous		1174	567	
heterogeneous		67	12082	

Table 4: Confusion matrix of the homogeneous-heterogeneous classification task. About 95% of the aggregates are correctly classified.

3.3 Sizes of primary WO₃ clusters and primary TiO₂ clusters

Figure 12a shows the difference between the network output $\text{CNN}^0(I(B))$ for the STEM images $I(B) = \{I_{i_1}, \ldots, I_{i_\nu}\}$ corresponding to the aggregates $A_{i_1}, \ldots, A_{i_\nu}$ in a batch B (given in Eq. (15), i.e., prior to rounding of the output, which would result in the estimator $\hat{\theta}_0$) and the preset value $\theta_0(B)$ of the model parameter θ_0 . Figure 12b shows the error distribution of $\hat{\theta}_0$ after rounding, where in about 48% of all cases the value of $\hat{\theta}_0$ coincides with θ_0 . Additionally, in more than 92% of the cases, the error of $\hat{\theta}_0$ is less than or equal to 1. Although the largest mean absolute error occurs in the case of $\theta_0 = 6$, the resulting inaccuracy corresponds to an average relative error of about 20%.

The quality of the estimator θ_1 introduced in Section 2.3.5 is similar to that of θ_0 , see Figure 13. After rounding the output of the network CNN¹, 32% of the predictions coincided with the preset values of θ_1 . In about 82% of the cases, an error less than or equal to 1 occurred. The mean absolute error for $\theta_1 = 6$ is equal to 1.44, where the resulting inaccuracy corresponds to an average relative error of about 24%.



Figure 12: Prediction error of $\hat{\theta}_0$. In Subfigure (a), the differences between the network output $\text{CNN}^0(I(B))$ and the preset values $\theta_0(B)$ of the model parameter θ_0 are shown. The prediction error of $\hat{\theta}_0$ after applying the rounding operation is displayed in Subfigure (b).



Figure 13: Prediction error of $\hat{\theta}_1$. In Subfigure (a), the differences between the network output $\text{CNN}^1(I(B))$ and the preset values $\theta_1(B)$ of the model parameter θ_1 are shown. The prediction error of $\hat{\theta}_1$ after applying the rounding operation is displayed in Subfigure (b).

3.4 Further structural descriptors of hetero-aggregates

Recall that the goal of the method presented in this paper is to generate realistic digital shadows of hetero-aggregates in 3D, solely from observations provided by 2D STEM images of the aggregates. For that purpose, the parameters $\theta_{D_f}, \theta_{\rho}, \theta_0, \theta_1$ of the stochastic 3D model introduced in Section 2.1.1 are predicted in order to specify the model configuration with which to generate digital shadows. However, so far, only the accuracy of the predictors $\hat{\theta}_{D_f}, \hat{\theta}_{\rho}, \hat{\theta}_0, \hat{\theta}_1$ for $\theta_{D_f}, \theta_{\rho}, \theta_0, \theta_1$ was evaluated, rather than investigating further structural descriptors of hetero-aggregates in order to evaluate the structural similarity between the resulting digital shadows and the original hetero-aggregates, i.e., the aggregates that were used for predicting the model parameters $\theta_{D_f}, \theta_{\rho}, \theta_0, \theta_1$. Moreover, many structural properties of the digital shadows are influenced by multiple model parameters, and thus, evaluating the quality of the four predictors $\hat{\theta}_{D_f}, \hat{\theta}_{\rho}, \hat{\theta}_0, \hat{\theta}_1$ separately is not sufficient. Therefore, three further structural descriptors, which characterize the 3D morphology of hetero-aggregates and have not yet been considered in this paper, are investigated in order to assess the similarity between original aggregates and corresponding digital shadows, see also Figure 1c-d.

3.4.1 Average cluster size and coordination numbers

The average cluster size $S_{\text{TiO}_2}(A)$ of TiO₂ particles of an aggregate $A = \{p_i = (x_i, r_i, l_i) : x_i \in \mathbb{R}^3, r_i \in \mathbb{R}^+, l_i \in \{0, 1\}, 1 \le i \le N\}$. describes the average cardinality of clusters of connected TiO₂ particles in A. It is given by

$$S_{\text{TiO}_{2}}(A) = \frac{1}{\#C_{\text{TiO}_{2}}(A)} \sum_{c \in C_{\text{TiO}_{2}}(A)} \#c,$$
(17)

where $C_{\text{TiO}_2}(A)$ denotes the set of all TiO₂ clusters in A. While the value of $S_{\text{TiO}_2}(A)$ is primarily influenced by the preset values of θ_1 and θ_{ρ} , the value of θ_0 also has some (minor) influence on $S_{\text{TiO}_2}(A)$ through its appearance in the definition of the Bernoulli-distributed labels L_k of the stochastic 3D model, see Section 2.1.1.

Furthermore, the so-called average hetero-coordination number $Z_{\text{hetero}}(A)$ of an aggregate A is considered, which is given by

$$Z_{\text{hetero}}(A) = \frac{1}{\#A} \sum_{p \in A} \#\{p' \in A \colon p, p' \text{ are in contact}, l \neq l'\}$$
$$= \frac{2\#\{\text{set of heterogeneous contacts in } A\}}{\#A},$$
(18)

where #A(=N) is the total number of particles in A. Thus, $Z_{\text{hetero}}(A)$ is the average number of contacts of particles in A with particles of the other material. Finally, the average coordination number $Z_{\text{total}}(A)$, given by

$$Z_{\text{total}}(A) = \frac{1}{\#A} \sum_{p \in A} \#\{p' \in A \colon p, p' \text{ are in contact}\}$$
$$= \frac{2\#\{\text{set of contacts in } A\}}{\#A}, \qquad (19)$$

is considered, which is the average number of contacts of particles in A to other particles, regardless of their material.

Since the number of contacts of a particle within an aggregate A strongly depends on the shape of A, the model parameter θ_{D_f} significantly influences the values of the descriptors $Z_{hetero}(A)$ and $Z_{total}(A)$. Further, $Z_{hetero}(A)$ tends to increase with θ_{ρ} close to 0.5 and decreasing primary cluster sizes determined by θ_0 and θ_1 .

3.4.2 Comparison of original hetero-aggregates and their digital shadows

To evaluate the quality of the predictor $\hat{\theta} = (\hat{\theta}_{D_f}, \hat{\theta}_{\rho}, \hat{\theta}_0, \hat{\theta}_1)$ in terms of the structural descriptors introduced in Section 3.4.1, 50 configurations of $\theta = (\theta_{D_f}, \theta_{\rho}, \theta_0, \theta_1)$ were selected at random, out of the index set *E* of evaluation data.

For each of these numerical specifications of θ , 800 new aggregates A_1, \ldots, A_{800} were drawn from the corresponding stochastic 3D model, and their structural descriptors $S_{\text{TiO}_2}(A_i)$, $Z_{\text{hetero}}(A_i)$ and $Z_{\text{total}}(A_i)$ for $i \in \{1, \ldots, 800\}$ were computed. Furthermore, for each case, the (preset) ground-truth parameter vector θ has been predicted using the methods explained in Section 2.3. Then, for each of the 50 specifications of $\hat{\theta}$, 800 additional aggregates A'_1, \ldots, A'_{800} and computed their structural descriptors $S_{\text{TiO}_2}(A'_i)$, $Z_{\text{hetero}}(A'_i)$ and $Z_{\text{total}}(A'_i)$ for $i \in \{1, \ldots, 800\}$ were generated. Figure 14 visualizes the distributions of these structural descriptors for four numerical specifications of θ , where the aggregates A_1, \ldots, A_{800} and A'_1, \ldots, A'_{800} were generated using either the preset parameter vector θ (blue) or its prediction $\hat{\theta}$ (orange), respectively.

Note that the gaps in the histograms of the average coordination numbers $Z_{\text{total}}(A_i)$ and $Z_{\text{total}}(A'_i)$ (right column) are due to the limited size of the considered aggregates, see Section 2.1.2. More specifically, the average coordination numbers $Z_{\text{total}}(A_i)$ and $Z_{\text{total}}(A'_i)$ given in Eq. (19), of aggregates A_i, A'_i with sizes smaller than or equal to 80, can only take values in the set

$$H = \{\frac{2q_1}{q_2} : q_1, q_2 \in \mathbb{N}, q_1 \le q_2 \le 80\},\tag{20}$$

where $H \cap (1.975, 2) = \emptyset$ because of the limited denominator q_2 on the right-hand side of Eq. (20).

Furthermore, note that the predictor $\hat{\theta}$ for θ displayed in the top row of Figure 14 has a much smaller mean absolute error than the one displayed in the second row. Nevertheless, the latter (blue and orange) histograms show a higher agreement than those in the top row of Figure 14. Meaning that a high degree of similarity (in terms of MAE) of θ and $\hat{\theta}$ does not necessarily imply a high degree of similarity of the resulting descriptor distributions.

We quantitatively analyzed this discrepancy between the distributions of the structural aggregate descriptors resulting from the preset configuration of model parameters and their prediction. For that purpose, the absolute difference of the means of these pairs of distributions were computed. For example, the mean values of $S_{\text{TiO}_2}(A_i)$ and $S_{\text{TiO}_2}(A_i')$ (vertical lines) in the top row of Figure 14 are equal to 5.38 and 11.00 for the preset parameter vector θ and its prediction $\hat{\theta}$, respectively. This results in an absolute error of 5.62. Over all 50 pairs of θ and $\hat{\theta}$, a MAE error of 2.165 is achieved, see also Table 5, where the MAEs for all three structural descriptors considered in this section are given. Even though the MAE is an easily interpretable metric for quantifying prediction errors, it is scale-dependent. Thus, comparing MAE values for structural descriptors, which belong to different length scales is impracticable. Unlike the MAE, the so-called coefficient of determination R^2 is scale-independent and thus an interpretable quantity for comparing the predictive power for different structural descriptors. In particular, R^2 is defined as

$$R^{2}(y, \widehat{y}) = 1 - \frac{\text{MSE}(y, \widehat{y})}{\text{MSE}(y, \overline{y})}.$$

Here, the vectors $y = (y_1, \ldots, y_{50}), \hat{y} = (\hat{y}_1, \ldots, \hat{y}_{50}) \in \mathbb{R}^{50}$ consist of the mean values of the distributions of the given aggregate descriptor computed for the 50 preset specifications of θ and their predictions $\hat{\theta}$. More precisely, for $j \in \{1, \ldots, 50\}, y_j = \frac{1}{800} \sum_{i=1}^{800} \gamma(A_{ij})$ and $\hat{y}_j = \frac{1}{800} \sum_{i=1}^{800} \gamma(A'_{ij})$, where γ stands for either $S_{\text{TiO}_2}, Z_{\text{hetero}}$ or Z_{total} , and A_{ij}, A'_{ij} denote the *i*-th aggregate drawn from the *j*-th specification of θ and its prediction $\hat{\theta}$, respectively. Furthermore, $\overline{y} = \frac{1}{50} \sum_{i=1}^{50} y_i$. The coefficient of determination R^2 takes values ranging from 0 to 1, where the value of R^2 can be interpreted as the fraction of variability in y explained by \hat{y} . The R^2 values for the considered structural descriptors are given in Table 5.



Figure 14: Distribution of the structural descriptors $S_{\text{TiO}_2}(A_i)$ and $S_{\text{TiO}_2}(A'_i)$ (left column), $Z_{\text{hetero}}(A_i)$ and $Z_{\text{hetero}}(A'_i)$ (middle column), as well as $Z_{\text{total}}(A_i)$ and $Z_{\text{total}}(A'_i)$ (right column) of the original aggregates A_i (blue) and their digital shadows A'_i (orange), for four numerical specifications of θ and their predictions $\hat{\theta}$. For computing the histograms, 20 equidistant bins have been employed, which span the entire range of respective values on the *x*-axis.

	S_{TiO_2}	$Z_{\rm hetero}$	$Z_{\rm total}$
MAE	2.165	0.056	0.007
\mathbb{R}^2	0.84	0.85	0.87

Table 5: Discrepancy between the mean values of the distributions of S_{TiO_2} , Z_{hetero} , and Z_{total} , with respect to the mean absolute error MAE and the coefficient of determination R^2 , computed for the 50 preset specifications of θ and their predictions $\hat{\theta}$.

4 Discussion

Before discussing the prediction quality of the proposed method for individual model parameters, the choice of the basic network architecture f is shortly investigated. As mentioned in Section 2.3, the network architecture of f can be easily exchanged to other architectures. Therefore, the performance of the proposed network architecture is compared with that of Inception-v3 and ResNet [20]. In Table 6 the prediction quality of these architectures regarding $D_{\rm f}$ are compared. It can be observed that the best results were achieved by our proposed network, thanks to its problem-specific optimized architecture. Therefore, this architecture serves as the foundation for all the results

in the present paper, which will be discussed in more detail later in this section.

The analysis of image data in order to determine the fractal dimension of finite aggregates has been a popular approach for some time. Two commonly used methods for this purpose are the box counting and sandbox methods, which are relatively simple image analysis tools [52, 8]. These methods can provide meaningful structural information, but the quality of the results is highly dependent on the quality of the images. Specifically, high contrast and resolution are necessary to obtain clear STEM images from which accurate structural information can be extracted. However, in cases where a high fractal dimension is present, i.e., $D_{\rm f}(A) > 2$, the accuracy of these classical methods decreases due to problems with geometric opacity, see [53]. Similarly, the prediction quality of the CNN-based method decreases for fractal dimensions $D_{\rm f} > 2$, see Figure 10a, but nevertheless the achieved accuracy for these cases is reasonably high. Furthermore, the CNN approach works well independently of the aggregate size, see Figure 16a. In order to further assess the performance of our predictor θ_{D_f} , the model parameter θ_{D_f} was predicted with a reference method from literature, namely, a method based on 2D box counting [8]. Therefore, the following steps have been performed: First analogously to the method described in [8], the box counting fractal dimension $D_{f,BC}(I)$ was computed for each STEM image I. Similarly to [8] we observe a deviation between $D_{f,BC}(I)$ and the value for $D_f(A)$ of the corresponding agglomerate A. Therefore, analogously to the method described in [8] we calibrated a correction step, i.e., using pairs of $D_{f,BC}$ and D_f values we fitted a power-law-based regression model. More precisely, the regression parameters a = 0.74 and b = 2.19 of

$$\widehat{D}_{\rm f}^{\rm BC}(I) = a D_{\rm f,BC}(I)^b \tag{21}$$

were fitted such that the MSE between the corrected fractal dimension $\widehat{D}_{\rm f}^{\rm BC}(I)$ and the true fractal dimension $D_{\rm f}(A)$ with respect to all pairs of aggregates A and corresponding STEM images I is minimized. Finally, the box counting-based prediction $\widehat{\theta}_{\rm D_f}^{\rm BC}$ of the model parameter $\theta_{\rm D_f}$ was computed analogously to $\widehat{\theta}_{\rm D_f}$, i.e.,

$$\widehat{\theta}_{\mathrm{D}_{\mathrm{f}}}^{\mathrm{BC}}(B) = \frac{1}{\nu} \sum_{j=1}^{\nu} \widehat{D}_{\mathrm{f}}^{\mathrm{BC}}(I_{i_j}), \qquad (22)$$

see Section 2.3.2 for more information. The errors of these predictions are displayed in Figure 15. A clear improvement of prediction quality can be observed, when considering the CNN-based method instead of the conventional box counting method, see Figures 10 and 15. The lower quality of the box counting-based prediction could be attributed to the following limitations: on the one hand, its inability to extract depth and mass information from STEM images, and on the other hand, its difficulty in accurately analyzing aggregates composed of a small number of primary particles.

Probably the most comparable conventional method for determining the mixing ratio of an aggregate via its 2D STEM image, is based on determining the particle label of each pixel using a threshold value. More specifically, depending on the pixel intensity, the pixel is classified as TiO_2 , WO_3 or background, and then, using the a priori known particle size distributions, a mixing ratio can be predicted. However, since the representation of thick TiO_2 particles or of many overlapping TiO_2 particles can have the same pixel intensity values as the representation of thin WO_3 particles, this threshold approach has a large source of errors [54]. The best appearing thresholds using a "brute force" algorithm on a representative data were determined. This results in an MAE of 0.078 per aggregate when estimating the mixing ratio. Compared to the MAE of 0.053, see Section 3.2, of the CNN approach described in the present paper, the error increases by 40% for the thresholding method described above. This is likely due to the increased values of pixel intensity caused by overlapping particles (see Figure 8), where these pixels with increased intensity values tend to be classified as WO_3 . Therefore, conventional threshold methods become increasingly inaccurate with an increasing number of overlapping particles, contrary to the behavior of the CNN approach proposed in the present paper, see Figure 16b.

Regarding the prediction of the remaining two model parameters θ_0 and θ_1 , as far as we know, there is no comparable conventional method based on 2D image data. Such methods, if they do not consider depth information,

would not be able to recognize if overlapping particles are touching or not, and thus, it is unlikely that they can accurately predict the values of θ_0 and θ_1 .

Recall that the objective of the present paper is to generate digital shadows that are stochastically equivalent to the ground-truth aggregates used for model fitting. These digital shadows, which have known a 3D structure, can then be employed to predict the structural properties of the ground-truth aggregates at significantly reduced costs. Therefore, rather than just evaluating the accuracy of the predicted model parameters $\hat{\theta} = (\hat{\theta}_{D_f}, \hat{\theta}_{\rho}, \hat{\theta}_0, \hat{\theta}_1)$, the morphological similarities of the resulting digital shadows and their ground truth in terms of further structural descriptors, i.e., average clusters sizes and coordination numbers were also investigated. As already mentioned in Section 3.4.2, the MAE of $\hat{\theta}$ is no appropriate tool to evaluate the similarity of digital shadows and their ground-truth aggregates. For instance, an extreme mixing ratio leads to a situation where the precision of either $\hat{\theta}_0$ or $\hat{\theta}_1$ has only a negligible impact on the structure of the resulting digital shadows is more strongly affected by small errors and rounding of $\text{CNN}^0(I(B))$ and $\text{CNN}^1(I(B))$ when the values of θ_0 and θ_1 are small, as opposed to when they are large. In particular, errors in the prediction of ground truths for small values of θ_0 and θ_1 result in higher relative errors. In such cases, large relative errors seem to have a greater impact on the structural discrepancies observed between aggregates generated for predicted and preset model parameters, see Figure 17. This effect can be further exacerbated by the application of subsequent rounding operations.

Although the predictor $\hat{\theta} = (\hat{\theta}_{D_f}, \hat{\theta}_{\rho}, \hat{\theta}_0, \hat{\theta}_1)$ proposed in this paper shows only minor discrepancies across all descriptors listed in Table 1, adapting the model and training process to address the issues mentioned above could enhance the similarity of digital shadows and original aggregates even further. For example, expanding the possible values of θ_0 and θ_1 to the interval [1, 6], rather than just considering the discrete set $\{1, 2, \ldots, 6\}$, would result in a diversity of aggregates, while also avoiding rounding errors that can arise in the prediction of θ_0 and θ_1 . More specifically, this could be achieved by modifying the aggregation model $\Psi_{N'}$ introduced in Eq. (7), such that the sizes of the primary clusters are randomly distributed, instead of choosing a constant cluster size. This would achieve a more detailed coverage of possible aggregate structures, especially for small values of θ_0 and θ_1 . The training of CNNs could benefit from an adapted cost function that takes the values of other model parameters into account and assigns weights to errors based on the importance of the ground truth to be predicted.

	our architecture	Inception-v3	ResNet
relative run time	1	3.8	5
MAE	0.0671	0.0958	0.1762

Table 6: Comparison of network architectures: In our experiments, our network architecture outperforms both the ResNet and Inception-v3 architectures in terms of run time and prediction quality. The displayed run time is normalized by the runtime of our architecture. The Mean Absolute Error (MAE) is computed for the predictions $\hat{D}_{\rm f}$ of $D_{\rm f}$, analogous to Figure 10a, for all three network architectures.



Figure 15: Prediction error of $\hat{D}_{\rm f}^{\rm BC}$ (left) and $\hat{\theta}_{\rm D_f}^{\rm BC}$ (right). In Subfigure (a) the quality of the prediction of the fractal dimension per aggregate is visualized, where the colors are computed by means of a Gaussian kernel density estimator. In Subfigure (b), the error regarding the prediction of the model parameter $\theta_{\rm D_f}$ is shown, where batches of size $\nu = 12$ are used for the computation of $\hat{\theta}_{\rm D_f}^{\rm BC}$.



Figure 16: Prediction error for fractal dimension (left) and mixing ratio (right), depending on the number of particles per aggregate, for aggregates taken from the evaluation data given by the index set E. The color of dots is chosen according to a 1D Gaussian kernel density estimator along the y-axis.



Figure 17: Distribution of the average heterogeneous coordination number $Z_{\text{hetero}}(A)$ for three different values of θ_0 . The dashed vertical lines show the mean value of $Z_{\text{hetero}}(A)$ for each of the three specifications of θ_0 , computed for a total of 2400 simulated aggregates A.

5 Conclusion

A novel neural network-based method has been developed in order to determine the parameters of a stochastic 3D model for synthetic hetero-aggregates, based on their 2D STEM images. Even though, the methodology has been derived and evaluated on the model system of TiO_2 -WO₃ hetero-aggregates, we believe that it can be transferred to further material systems analogously, simply by adapting the simulation algorithm for STEM images described in Section 2.1.3. This neural network-based approach has several advantages in hetero-aggregate analysis: (i) It is capable of fitting a (parametric) stochastic-geometry model solely based on 2D image data. (ii) The predicted model parameters exhibit relatively small errors. (iii) Through the use of synthetic data for the neural network training, there are no additional costs for acquiring real training data. (iv) By fitting the stochastic-geometry model instead of predicting individual descriptors, the entire probability distribution of the 3D morphology of aggregates is known, allowing for the computation of further hetero-aggregate properties, which have not been considered by the neural networks during training, see Section 4. (v) Realizations of the fitted stochastic-geometry model can be used as input for numerical simulations and systematic optimization of aggregate structures.

Note that the method has been calibrated using model realizations of the stochastic-geometry model described in [33], see also Section 2.1.2. Thus, the application of the trained networks to experimentally acquired STEM images can only be done if we can assume that the probability distribution of imaged hetero-aggregates can be described by our considered stochastic-geometry model. Otherwise, we have to adapt the model and its parameter space for these types of aggregates, e.g., if primary particles follow another size distribution. Once an adequate stochastic-geometry model is identified, the presented methodology can be applied analogously.

The prediction of certain material-specific parameters (e.g., the mixing ratio) can, in general, become more difficult if the materials exhibit similar pixel value intensities in STEM images. The model system of hetero-aggregates considered in the present paper exhibits a good material contrast in STEM images. However, only spherical particles were used for the generation of synthetic 3D aggregates. In future work, we will investigate the effectiveness of the proposed method for hetero-aggregates consisting of particles that feature similar STEM intensities but differ significantly in their shape or size. Moreover, since experimentally measured aggregates feature more varied cluster sizes than synthetically generated ones, it can be presumed that a larger variability in cluster sizes would require more comprehensive data sets in order to make accurate predictions, but investigating this effect systematically is still important. Finally, in a forthcoming study, the presented method will be experimentally validated. More precisely, experimentally acquired 3D STEM image data of hetero-aggregates will be analyzed to investigate how well the stochastic 3D model proposed in the present paper can describe real aggregates.

6 Acknowledgements

This work was financially supported by the German Research Foundation (DFG) through the research grants RO 2057/17-1, MA 3333/25-1 and SCHM 997/42-1.

References

- J. A. Pinedo-Escobar, J. Fan, E. Moctezuma, C. Gomez-Solís, C. J. Carrillo Martinez, and E Gracia-Espino. Nanoparticulate double-heterojunction photocatalysts comprising TiO₂(Anatase)/WO₃/TiO₂(Rutile) with enhanced photocatalytic activity toward the degradation of methyl orange under near-ultraviolet and visible light. ACS Omega, 6(18):11840–11848, 2021.
- [2] Y. Tae Kwon, K. Yong Song, W. In Lee, G. Jin Choi, and Y. Rag Do. Photocatalytic behavior of WO₃-loaded TiO₂ in an oxidation reaction. *Journal of Catalysis*, 191(1):192–199, 2000.
- [3] X. Yan, X. Zong, G. Q. Lu, and L. Wang. Ordered mesoporous tungsten oxide and titanium oxide composites and their photocatalytic degradation behavior. *Progress in Natural Science: Materials International*, 22(6):654–660, 2012.
- [4] J. Low, J. Yu, M. Jaroniec, S. Wageh, and A. A. Al-Ghamdi. Heterojunction photocatalysts. Advanced Materials, 29(20):1601694, 2017.
- [5] P.A. Midgley and M. Weyland. 3D electron microscopy in the physical sciences: the development of Z-contrast and EFTEM tomography. *Ultramicroscopy*, 96(3):413 431, 2003.
- [6] P. Gilbert. Iterative methods for the three-dimensional reconstruction of an object from projections. Journal of Theoretical Biology, 36(1):105 – 117, 1972.
- [7] S. Buchheiser, F. Kistner, F. Rhein, and H. Nirschl. Spray flame synthesis and multiscale characterization of carbon black-silica hetero-aggregates. *Nanomaterials*, 13(12):1893, 2023.
- [8] R. Wang, A. K. Singh, S. R. Kolan, and E. Tsotsas. Fractal analysis of aggregates: Correlation between the 2D and 3D box-counting fractal dimension and power law fractal dimension. *Chaos, Solitons & Fractals*, 160:112246, 2022.
- [9] M. Frei and F. E. Kruis. Image-based size analysis of agglomerated and partially sintered particles via convolutional neural networks. *Powder Technology*, 360:324–336, 2020.
- [10] C. Mahr, J. Stahl, B. Gerken, V. Baric, M. Frei, F. F. Krause, T. Grieb, M. Schowalter, T. Mehrtens, E. Kruis, L. Mädler, and A. Rosenauer. Characterization of mixing in nanoparticle hetero-aggregates by convolutional neural networks. *Nano Select*, 2024:2300128, 2024.
- [11] S. Chiu, D. Stoyan, W. Kendall, and J. Mecke. Stochastic Geometry and Its Applications. J. Wiley & Sons, 3rd edition, 2013.
- [12] S. Kench and S. J. Cooper. Generating three-dimensional structures from a two-dimensional slice with generative adversarial network-based dimensionality expansion. *Nature Machine Intelligence*, 3(4):299–305, 2021.

- [13] I. Goodfellow, Y. Bengio, and A. Courville. Deep Learning. MIT Press, 2016.
- [14] G. James, D. Witten, T. Hastie, and R. Tibshirani. An Introduction to Statistical Learning. Springer, 2nd edition, 2021.
- [15] T. Kirstein, L. Petrich, R. R. P. Purushottam Raj Purohit, J. Micha, and V. Schmidt. CNN-based laue spot morphology predictor for reliable crystallographic descriptor estimation. *Materials*, 16:3397, 2023.
- [16] K. Simonyan and A. Zisserman. Very deep convolutional networks for large-scale image recognition. arXiv preprint arXiv:1409.1556, 2014.
- [17] S. Mascarenhas and M. Agarwal. A comparison between vgg16, vgg19 and resnet50 architecture frameworks for image classification. In 2021 International Conference on Disruptive Technologies for Multi-Disciplinary Research and Applications (CENTCON), volume 1, pages 96–99. IEEE, 2021.
- [18] C. Szegedy, V. Vanhoucke, S. Ioffe, J. Shlens, and Z. Wojna. Rethinking the inception architecture for computer vision. In *Proceedings of the IEEE Conference on Computer Vision and Pattern Recognition (CVPR)*, pages 2818–2826, 2016.
- [19] K. He, X. Zhang, S. Ren, and J. Sun. Deep residual learning for image recognition. In Proceedings of the IEEE Conference on Computer Vision and Pattern Recognition (CVPR), pages 770–778, June 2016.
- [20] A. Khan, A. Sohail, U. Zahoora, and A. S. Qureshi. A survey of the recent architectures of deep convolutional neural networks. *Artificial Intelligence Review*, 53:5455–5516, 2020.
- [21] M. Neumann, S. E. Wetterauer, M. Osenberg, A. Hilger, P. Gräfensteiner, A. Wagner, N. Bohn, J. R. Binder, I. Manke, T. Carraro, and V. Schmidt. A data-driven modeling approach to quantify morphology effects on transport properties in nanostructured NMC particles. *International Journal of Solids and Structures*, 280:112394, 2023.
- [22] M. Weber, A. Grießer, E. Glatt, A. Wiegmann, and V. Schmidt. Modeling curved fibers by fitting R-vine copulas to their frenet representations. *Microscopy and Microanalysis*, 29(1):155–165, 2023.
- [23] M. Neumann, J. Staněk, O. M. Pecho, L. Holzer, V. Beneš, and V. Schmidt. Stochastic 3D modeling of complex three-phase microstructures in sofc-electrodes with completely connected phases. *Computational Materials Science*, 118:353–364, 2016.
- [24] B. Priffing, M. Neumann, D. Hlushkou, C. Kübel, U. Tallarek, and V. Schmidt. Generating digital twins of mesoporous silica by graph-based stochastic microstructure modeling. *Computational Materials Science*, 187:109934, 2021.
- [25] K. Giannis, C. Thon, G. Yang, A. Kwade, and C. Schilde. Predicting 3D particles shapes based on 2D images by using convolutional neural network. *Powder Technology*, 432:119122, 2024.
- [26] K. Fu, J. Peng, Q. He, and H. Zhang. Single image 3D object reconstruction based on deep learning: A review. *Multimedia Tools and Applications*, 80:463–498, 2021.
- [27] S. R. Forrest and T. A. Witten. Long-range correlations in smoke-particle aggregates. Journal of Physics A: Mathematical and General, 12(5):L109, 1979.
- [28] P. Meakin. Formation of fractal clusters and networks by irreversible diffusion-limited aggregation. *Physical Review Letters*, 51:1119–1122, 1983.

- [29] J. Cai, N. Lu, and C. M. Sorensen. Analysis of fractal cluster morphology parameters: Structural coefficient and density autocorrelation function cutoff. Journal of Colloid and Interface Science, 171(2):470–473, 1995.
- [30] M. L. Eggersdorfer and S. E. Pratsinis. The structure of agglomerates consisting of polydisperse particles. Aerosol Science and Technology, 46(3):347–353, 2012.
- [31] M. L. Eggersdorfer and S. E. Pratsinis. Agglomerates and aggregates of nanoparticles made in the gas phase. Advanced Powder Technology, 25(1):71–90, 2014.
- [32] A.V. Filippov, M. Zurita, and D.E. Rosner. Fractal-like aggregates: Relation between morphology and physical properties. Journal of Colloid and Interface Science, 229(1):261–273, 2000.
- [33] V. Baric, H. K. Grossmann, W. Koch, and L. M\u00e4dler. Quantitative characterization of mixing in multicomponent nanoparticle aggregates. Particle & Particle Systems Characterization, 35(10):1800177, 2018.
- [34] R. Jullien, M. Kolb, and R. Botet. Aggregation by kinetic clustering of clusters in dimensions d > 2. Journal de Physique Lettres, 45(5):211–216, 1984.
- [35] D. Van Dyck. Is the frozen phonon model adequate to describe inelastic phonon scattering? Ultramicroscopy, 109(6):677 – 682, 2009.
- [36] A. Rosenauer and M. Schowalter. STEMSIM a new software tool for simulation of STEM HAADF Z-contrast imaging. In A.G. Cullis and P.A. Midgley, editors, *Microscopy of Semiconducting Materials 2007*, volume 120 of *Springer Proceedings in Physics*, pages 170–172. Springer, 2008.
- [37] R. Diehl, G. Brandt, and E. Salje. The crystal structure of triclinic WO₃. Acta Crystallographica Section B, 34(4):1105–1111, 1978.
- [38] M. Horn and C. P. Schwerdtfeger. Refinement of the structure of anatase at several temperatures. Zeitschrift für Kristallographie, 136(3-4):273–281, 1972.
- [39] B. O. Loopstra and H. M. Rietveld. Further refinement of the structure of WO₃. Acta Crystallographica Section B, 25(7):1420–1421, 1969.
- [40] K. Sugiyama and Y. Takéuchi. The crystal structure of rutile as a function of temperature up to 1600°C. Zeitschrift für Kristallographie - Crystalline Materials, 194(1-4):305–314, 1991.
- [41] E. J. Kirkland. Advanced Computing in Electron Microscopy. Springer, 1998.
- [42] F. F. Krause, M. Schowalter, T. Grieb, K. Müller-Caspary, T. Mehrtens, and A. Rosenauer. Effects of instrument imperfections on quantitative scanning transmission electron microscopy. *Ultramicroscopy*, 161:146–160, 2016.
- [43] L. Jones and P. D. Nellist. Identifying and correcting scan noise and drift in the scanning transmission electron microscope. *Microscopy and Microanalysis*, 19(4):1050–1060, 05 2013.
- [44] K. Kumar Pal and K. S. Sudeep. Preprocessing for image classification by convolutional neural networks. In 2016 IEEE International Conference on Recent Trends in Electronics, Information & Communication Technology (RTEICT), pages 1778–1781, 2016.
- [45] P.Y. Simard, D. Steinkraus, and J.C. Platt. Best practices for convolutional neural networks applied to visual document analysis. In *Seventh International Conference on Document Analysis and Recognition*, pages 958–963, 2003.

- [46] C. F. G. D. Santos and J. P. Papa. Avoiding overfitting: A survey on regularization methods for convolutional neural networks. ACM Computing Surveys, 54:213, 2022.
- [47] H. Qassim, A. Verma, and D. Feinzimer. Compressed residual-vgg16 cnn model for big data places image recognition. In 2018 IEEE 8th Annual Computing and Communication Workshop and Conference (CCWC), pages 169–175. IEEE, 2018.
- [48] S. Ioffe and C. Szegedy. Batch normalization: Accelerating deep network training by reducing internal covariate shift. In *International Conference on Machine Learning*, pages 448–456. PMLR, 2015.
- [49] D.P. Kingma and J. Ba. Adam: A method for stochastic optimization. In Y. Bengio and Y. Le Cun, editors, Proceedings of the 3rd International Conference on Learning Representations, 2015.
- [50] T. Szandała. Review and comparison of commonly used activation functions for deep neural networks. arXiv preprint arXiv:2010.09458, 2020.
- [51] R. Mohammed, J. Rawashdeh, and M Abdullah. Machine learning with oversampling and undersampling techniques: Overview study and experimental results. In 11th International Conference on Information and Communication Systems (ICICS), pages 243–248, 2020.
- [52] G.C. Bushell, Y.D. Yan, D. Woodfield, J. Raper, and R. Amal. On techniques for the measurement of the mass fractal dimension of aggregates. Advances in Colloid and Interface Science, 95(1):1–50, 2002.
- [53] F. Maggi and J. C. Winterwerp. Method for computing the three-dimensional capacity dimension from twodimensional projections of fractal aggregates. *Physical Review E*, 69:011405, 2004.
- [54] B. Gerken, C. Mahr, J. Stahl, T. Grieb, M. Schowalter, F. F. Krause, T. Mehrtens, L. Mädler, and A. Rosenauer. Material discrimination in nanoparticle hetero-aggregates by analysis of scanning transmission electron microscopy images. *Particle & Particle Systems Characterization*, 40(9):2300048, 2023.