3D analysis of equally X-ray attenuating mineralogical phases utilizing a correlative tomographic workflow across multiple length scales

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Abstract

A correlative 3D characterization workflow by micro and nano X-ray computed tomography (Micro-CT, Nano-CT) and analytical scanning electron microscopy (SEM) is presented over different length scales for particle composite materials that apply to any powder at the size scale between 0.3 µm to 15 µm. In this case study, an artificial compound of calcite, talcum, dolomite, and magnesite providing constituent particles with similar morphology, size distribution, and chemical composition for multidimensional separation processes is analyzed. First, Micro-CT characterizes the particle morphology and distribution of a larger amalgamated volume. Then, a smaller, site-specifically prepared pillar is imaged by Nano-CT allowing for correlative investigations at higher-resolution. Afterwards, the Nano-CT reconstruction is informed slicewise by analytical SEM distinguishing particles with different chemical composition. The statistical interpretation of our results is improved by advanced post-processing and multidimensional analysis, allowing for quantitative characterization of the particles' size, phase distribution, and mineral degree of liberation.

Keywords: Multidimensional particle characterization, multiscale X-ray tomography, correlative 3D analysis, statistical image analysis, parametric copula

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1. Introduction

High-grade ore deposits become increasingly rare and 2 the existing ones are more and more depleted. This ten-3 dency is not new [1] but is clearly intensifying with the 4 increasing need for natural resources [2]. The pressure 5 to mine ores that are significantly more complex in their 6 structure than easily accessible material of the same qual-7 ity is rising. This means that a greater variety of grain 8 size and mineralogical neighboring phases occur in one 9 single particle. At the same time, the requirements for 10 the final product remain the same or even increase. In 11 this regard, standard separation methods reach their lim-12 32 its or even fail if the particle properties are very similar 13 22 to each other and separation is hardly or no longer possi-14 ble. Moreover, complexity increases due to a multitude of 15 new and more accurate measurement methods such as ad-16 vanced microbeam techniques, which are ideally brought 17 into correlation with each other [3]. In this context, three-18 38 dimensional (3D) measurement routines become more and 19

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more important, since they enable the precise analysis of complex particle morphologies and 3D spatial distribution at the same time.

Besides morphological and chemical composition, the steadily decreasing particle size brings up further challenges. In addition to traditional particle size characterization methods like sieving or laser diffraction measurements, which are always dependent on given class size or specific model assumptions to compute distributions from raw data, direct imaging methods are widely used to reveal multiple particle characteristics at once. Static two-dimensional (2D) imaging methods, such as light microscopy, were supplemented by dynamic methods such as dynamic light scattering [4] representing a significant development, especially with regard to the statistical representativeness of the samples [5]. Additional image tracking algorithms were used to compensate for the stereological bias [6] from the 2D image description, but only down to particle sizes of around $100 \,\mu\text{m}$ [7]. With these methods, a comprehensive 3D description of particle collectives smaller than 10 µm, as we focus on in this case study, is not possible. Thus, the extension to direct 3D measurement methods is needed to acquire morphological characteristics as well.

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43 During the last three decades, X-ray computed tomog-101 44 raphy (CT) has become a standard tool in many research¹⁰² 45 areas, like materials science [8] and geoscience [9, 10]. In₁₀₃ 46 particle technology, in 1992 first analyses were conducted₁₀₄ 47 in mineral processing on a limited number of particles with105 48 sizes down to 100 µm or in the non-particle-discrete anal-106 49 vsis of particulate suspensions in separators, e.g., with a¹⁰⁷ 50 hydrocyclone [11, 12]. These measurements are typically¹⁰⁸ 51 based on absorption contrast where the local X-ray at-109 52 tenuation in projection is detected depending on the local¹¹⁰ 53 sample thickness and material composition (X-ray atten-111 54 uation coefficients), so that existing phases with sufficient¹¹² 55 difference in atomic number can be discriminated [13, 14].¹¹³ 56 To be able to reconstruct the sample in 3D, many projec-114 57 tions under different viewing angles need to be acquired.115 58 Therefore, the sample in the stable beam path is typi-116 59 cally rotated by the sample stage of the CT instrument¹¹⁷ 60 within a tilt-angle range of at least 180° using a prede-118 61 fined tilt increment. There are several beam geometries119 62 available, two of which are used in the present study: a120 63 (quasi) monochromatic parallel X-ray beam in Nano-CT₁₂₁ 64 and a cone-shaped polychromatic X-ray beam in the case₁₂₂ 65 of Micro-CT. [15]. In both cases, the photon intensities on₁₂₃ 66 the detector are translated into gray-scale value images.124 67 The series of projection images is then used to reconstruct₁₂₅ 68 the 3D volumetric image by applying a mathematical re-126 69 construction algorithm [15]. In absorption contrast, the₁₂₇ 70 gray values in the reconstructed volumes resemble the lo-128 71 cal attenuation coefficients, which can be utilized using 3D₁₂₉ 72 image processing routines to identify and segment differ-130 73 ent phases and extract valuable morphological and chemi-131 74 cal information [15]. This kind of intuitive data analysis is₁₃₂ 75 not possible in the case of a polychromatic X-ray beam or₁₃₃ 76 insufficient contrast due to similar attenuation coefficient134 77 of the constituent phases of the sample, so that additional135 78 information for a robust segmentation is required. 79 136

In the case of powders with particles consisting of dif-137 80 ferent phases, another key parameter besides the particle-138 81 discrete information is the phase-discrete information. The139 82 combination of both can offer a detailed characterization₁₄₀ 83 of the particle and batch composition. It is noteworthy₁₄₁ 84 to say that a particle may consist of several grains, which₁₄₂ 85 in this context refers to a volume containing a single min-143 86 eral phase. In the field of mineral processing, one distin-144 87 guishes, for example, between valuable grains (or phases)145 88 and non-valuable grains within particles to ascertain how146 89 well the valuable materials can be enriched. The volume₁₄₇ 90 ratio of all particles, which consist of only the valuable148 91 phase, to the total of all particles, which contain this spe-149 92 cific phase is called liberation degree (LD). The LD is150 93 an important but aggregated parameter for the process-151 94 ing (e.g., milling or separation processes) of primary and 152 95 secondary mineral raw materials. In the present study, the153 96 3D particle-discrete determination of the LD will be shown₁₅₄ 97 as a concrete application example for the use of the corre-155 98 lated particle-discrete and grain-discrete data. In contrast₁₅₆ 99

to existing studies [16], this is demonstrated here using an example of X-ray attenuating mineral phases with comparable X-ray attenuation contained in particles smaller than 10 µm.

A standard measurement set-up for a 2D determination of the mineralogical composition on polished epoxyembedded sections, i.e. the mineral liberation analysis (MLA) [17], consists of scanning electron microscopy (SEM) using the back-scattered electron (BSE) signal in combination with an energy dispersive X-ray spectroscopy (EDXS) detector for elemental analysis. A computer software in combination with databases on known materials can now be used to identify the containing minerals and provide a color image, where each color corresponds to a certain mineral. It is typically difficult to correct for the discrepancy in analysis results of 2D sectional images compared to results obtained by the analysis of real 3D structures, called stereological bias [6]. Even with an optimized sample preparation strategy to suppress segregation effects due to sedimentation [18, 19], the influence of the particle's internal structure is significant [20]. With the presented 3D methodology, it will be possible to discretely perform this liberation analysis of multiphase particle systems. Previous studies have done this, e.g., for binary systems [21], or for other particle size scales of around $100 \,\mu m$ [16].

The shape of particles is often correlated with the mineralogical composition of particles [22]. In some cases, this allows for a mineralogical characterization by means of particle shape characteristics, which can easily be determined from CT data. However, if distributions of 3D morphological particle descriptors overlap, there is a need for additional phase-specific information to guarantee a valid identification, especially in terms of mineralogical phases. Already established workflows can be used (i) to distinguish between phases with strongly differing attenuating properties, e.g., gold phase in other mineralogical components. Here, the qualitative difference is clearly visible in the gray value histogram and can be used for direct phase identification [23]. Another workflow is (ii) to use characteristic, element-specific absorption properties, where a sample is scanned with two different energies slightly above and below a phase specific discontinuity in the absorption behavior, e.g., a X-ray absorption K-edge of the elements in one known phase of interest [24]. A relatively new workflow is (iii) to use an additional energy-dispersive detector, called spectral CT, which is able to distinguish between multiple phases at once [25].

The alternative, new workflow proposed in the present study possesses the following advantages in comparison to the existing workflows (i)-(iii) mentioned above. It enables to distinguish between mineral components of a micronsized particle system having comparable X-ray attenuating properties and consisting of more than one phase. Furthermore, our workflow can be implemented with existing lab-based CT devices without the need for an extension by an additional detector. Note that our workflow uses additional prior information regarding local chemi-



Figure 1: Particle systems considered in this study, (left) talcum with three different sub-phases of dolomite, magnesite, and talcum; (right) saxolite.

Table 1: Chemistry and density of the two particle systems saxolite and talcum, the latter consists of three different sub-phases, which are dolomite, magnesite and talcum. Attenuation at 5.4 keV represents the attenuation length within a given material of the quasimonochromatic Nano-CT beam.

	Chemistry	Density in $g \cdot cm^{-3}$	$\begin{array}{c} \text{Attenuation} \\ \text{at} \ 5.4 \mathrm{keV} \end{array}$
Saxolite Talcum	$CaCO_3$	2.75	$15.3\mu\mathrm{m}$
Dolomite	$MgCa(CO_3)_2$	2.86	$23.5\mu{ m m}$
Magnesite	$MgCO_3$	3.00	$55.4\mathrm{\mu m}$
Talcum	$Mg_3Si_4O_{10}(OH)_2$	2.76	$34.7\mu\mathrm{m}$

cal composition from EDXS, which was performed in a₁₉₈ 157 similar way in Furat et al. [26] covering particle sizes of₁₉₉ 158 $315\,\mu\mathrm{m}$ to $500\,\mu\mathrm{m}$. In the present study, their method₂₀₀ 159 is extended to particle sizes below 10 µm by correlating₂₀₁ 160 Micro-CT measurements of medium-resolution $(\sim 700 \text{ nm})_{202}$ 161 with high-resolution ($\sim 150 \,\mathrm{nm}$) measurements, which are₂₀₃ 162 conducted using a combination of Nano-CT, SEM imag-204 163 ing and SEM-EDXS, where specific slices in the Nano-205 164 CT volume are accessed using focused ion beam (FIB)₂₀₆ 165 milling. This correlative measurement routine enables ex-207 166 aminations over multiple length scales with a consistent₂₀₈ 167 and adaptable analysis and preparation workflow. 168 209

The paper is structured as follows. In Sec. 2, we shortly₂₁₀ 169 introduce the considered particle system, describe the mea-211 170 surement workflow for volume correlation followed by the₂₁₂ 171 applied image processing strategies and data analysis meth-213 172 ods. In Sec. 3, we present the results of the correlation be- $_{214}$ 173 tween the Nano-CT volumes and the cut sections from the₂₁₅ 174 FIB-SEM-EDXS analysis, and the phase determination for₂₁₆ 175 materials characterization. In Sec. 4, we discuss the results₂₁₇ 176 with respect to the precision of the derived quantitative₂₁₈ 177 data as well as its reliability concerning a larger scale. We₂₁₉ 178 then close the loop to practical applications by discussing₂₂₀ 179 the benefits of our correlative workflow to tackle the chal-221 180 lenges of separation processes. In Sec. 5, we summarize the₂₂₂ 181 advantages of the presented workflow in terms of statistical₂₂₃ 182 significance and the 3D particle-discrete and grain-discrete₂₂₄ 183 184 provision of the analysis results compared to conventional 2D methods. 185 225

186 2. Materials and Methods

187 2.1. Particle System

The materials considered in this study are natural prod-²³⁰ ucts and therefore differ in their purity and homogene-²³¹ ity. Saxolite, type Saxolith[®]2 extra, was received from²³² the Erzgebirgische Kalkwerke GmbH GEOMIN, Germany.²³³ Talcum was received from Giessereitechnik Wystrach GmbH⁴ ¹⁹³ & Co. KG, Germany. Exemplary SEM images of these²³⁵ materials are shown in Fig. 1.²³⁶

The corresponding particle size distributions of saxo-²³⁷ lite determined by laser diffraction and of talcum parti-²³⁸ cles determined by sedimentation analysis can be found in²³⁹ the Supplementary material. The distributions overlap in large parts between 1 μ m to 10 μ m in diameter, showing that particle size alone is not fully suitable as a separating characteristic. As it can be seen in Tab. 1, also the densities do not differ significantly. Considering talcum as a mixture with a mean density from the tabulated values for dolomite, magnesite and talcum, this becomes even clearer. Additionally, some saxolite particles are composites which contain traces of talcum (up to 4 %), making it even harder to identify these materials. Assuming comparable particle sizes, a distinction between the two phases on the basis of the gray-scale histograms from both Micro-CT and Nano-CT measurements is not possible.

For the creation of a suitable and valid analysis workflow, a mixture of both particle systems (talcum and saxolite) was manufactured. A mixing ratio of 30 to 70 (volume fraction) was chosen as an expected scenario for a future application of the analysis workflow, in our case, the characterization of a multidimensional separation with respect to two (including the sub-phases, four) particle characteristics. The separation itself is realized by a combination of a deflector wheel classifier (separation by size) and a triboelectric charging process due to particle wall collisions (separation by charge) [27]. The various mineral components carry different charges due to their triboelectric characteristics, e.g., their electrochemical potential, so that they can be separated by their electrical mobility.

2.2. Measurement Workflow for Volume Correlation

The following workflow is a correlative study across different X-ray and electron microscopes. Every microscope adds information by utilizing its advantages of the capable sample size, achievable resolution and information content (morphological, chemical). The scale bridging nature is necessary to determine the coordinates for further processing with the help of laser ablation and to ensure the representativeness of the individual volumes of high-resolution scans using Nano-CT. The low-resolution measurements performed with Micro-CT are used to determine the homogeneity of the particle dispersion of the initial sample and to check its suitability (absence of cracks and air bubbles) for the subsequent sample preparation workflow by laser ablation and FIB-SEM.

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2.2.1. Sample Preparation 240

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To guarantee a well-dispersed homogeneous sample, 293 241 the particles are mixed with low X-ray attenuating carbon₂₉₄ 242 black nanoparticles acting as spacers. Touching particles²⁹⁵ 243 are a potential source for errors in the image processing₂₉₆ 244 workflow when separating to create particle-discrete data²⁹⁷ 245 sets, which are essential for further quantitative analy-298 246 For a detailed description of the sample prepara-299 ses. 247 tion method, see Ditscherlein et al. [19]. The epoxy-carbon₃₀₀ 248 black matrix is well machinable, using mechanical prepa-301 249 ration techniques as well as using high-energy radiation₃₀₂ 250 techniques, e.g., laser ablation to create sample volumes³⁰³ 251 for Nano-CT experiments and FIB to cut sections for cor-304 252 relative SEM measurements. 253 305

2.2.2. Micro-CT Measurements 254

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Micro-CT scans were performed using a Zeiss Xradia³⁰⁸ 255 510 Versa, with a polychromatic X-ray source, a rotat-309 256 ing tungsten target, a maximum acceleration voltage of³¹⁰ 257 160 keV, and a maximum power of 10 W. Compared to³¹¹ 258 conventional Micro-CT systems, where the magnification³¹² 259 is determined by the geometrical arrangement of source,³¹³ 260 sample and detector, an additional magnifying optic (fac-314 261 tors 0.4x, 4x, 20x, 40x) allows a minimum voxel size of³¹⁵ 262 0.3 µm. The Micro-CT scans were performed on a man-³¹⁶ 263 ually cut bar (see Sec. 2.2.1) using two different magni-317 264 fications (see Tab. S2) – one lower resolution overview³¹⁸ 265 scan to check the Micro-CT sample for possible prepara-319 266 tion artifacts, and one medium resolution scan to confirm³²⁰ 267 the representativeness and homogeneity of the particle dis-321 268 persion and to obtain the coordinates for the subsequent³²² 269 high-resolution Nano-CT scans, see Fig. 2. Since the ap-270 plied preparation method cannot completely prevent the³²³ 271 incorporation of air bubbles into the matrix, bubble-free³²⁴ 272 regions can also be identified here that are suitable for fur-³²⁵ 273 ther preparation with the laser ablation system to create³²⁶ 274

2.2.3. Laser Ablation 276

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329 After the Micro-CT experiments, the sample size was³³⁰ 277 further reduced for Nano-CT investigations using a 3D-³³¹ 278 Micromac microPREP[™] PRO laser ablation system. The³³² 279 sample top region was ablated down from a manually cut³³³ 280 bar of 400 μ m to a pillar with 60 μ m in diameter, see Fig. S3,³³⁴ 281 which fits into the field of view (FOV) of the Nano-CT in-³³⁵ 282 strument. For both Micro-CT and Nano-CT machines,336 283 which were used in this work, the detectors have a cer-³³⁷ 284 tain number of pixels and therefore a static FOV for each³³⁸ 285 pre-defined magnification setting. This leads to a fixed³³⁹ 286 captured volume, where the sample size should match the³⁴⁰ 287 FOV at best to avoid reconstruction artifacts due to inte-³⁴¹ 288 rior tomography restrictions [28]. Along the length of the³⁴² 289 pillar, the FOV can easily extended by acquiring multiple³⁴³ 290 tomographic tilt series along the vertical axis. 344 291

cylinders for the high-resolution Nano-CT scans.

2.2.4. Nano-CT Measurements

High-resolution 3D scans were performed using a Zeiss Xradia 810 Ultra Nano-CT instrument, which operates with quasi-monochromatic X-rays at constant photon energy with parallel beam geometry. The quasi-monochromatic beam is a result of the filtering properties by the microscope's X-ray optics containing a condenser and a Fresnel zone plate lens that attenuate certain wavelengths due to their dispersion efficiency. The lenses are optimized to transmit an even narrower bandwidth of the characteristic X-rays. In this case, X-rays with an energy of 5.4 keV are used corresponding to the characteristic X-ray energy of the K α -line of a rotating chromium anode. We utilize absorption contrast mode which exhibits mainly massthickness contrast imaging with a minimum voxel size of 64 nm. The quasi-monochromatic X-rays allow a direct correlation of the reconstructed image intensities to local attenuation coefficients, which manifests itself in the different brightness of the particles in slices of the reconstructed tomograms. Volume reconstruction based on simultaneous iterative reconstruction technique (SIRT) was performed with an in-house Python script based on the algorithms provided by the ASTRA toolbox [29, 30, 31]. The highresolution scans are needed to acquire the 3D particlediscrete information of the whole volume, which is a combination of three vertically stacked individual volumes (see Fig. S4(b) for the positions of the single Nano-CT reconstructions indicated in the complete, stitched Nano-CT volume). Detailed measurement parameters of the tomographic measurements can be found in the supplementary material table S2.

2.2.5. FIB Preparation, SEM Imaging and SEM-EDXS

FIB milling, SEM imaging and energy dispersive Xray spectroscopy (EDXS) were conducted in a FEI Helios NanoLab 660 SEM/FIB dual-beam system. For the FIB milling process, which is used for the preparation of thinner and finer Nano-CT sample pillars, Ga+ ions with high kinetic energy are directed onto the sample surface to achieve local material ablation. The 60 µm pillar was cut vertically along the rotation axis in order to remove the melting zone of the laser ablation. As described in Section 2.2.4, the sample has been imaged in 3D at three positions by acquiring Nano-CT tilt series (see Fig. S4(b) for the slice positions indicated in the complete, stitched Nano-CT volume). Thus, the sample was sliced in the regions of each of these three corresponding Nano-CT data sets with 28° horizontal inclination (limited by the SEM/FIB stage) and imaged after each cut. Charging effects during the milling process are common with these minerals. Thus, each cut is processed in a single run. A connected secondary electron (SE) detector provides the image data of the current sectional plane surface. Backscattered electron (BSE) imaging and EDXS of the same cut surfaces make it possible to distinguish between the talcum sub-phases and saxolite, by providing additional information about the local chemical composi-

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Figure 2: Mechanically dispersed particle system embedded in an epoxy-carbon black matrix, see [19] for a detailed description of the preparation, (a) glued on a needle pin, scanned with low resolution by Micro-CT, represented here by (b) a side view and (c) an exemplary horizontal slice through the reconstructed volume (d) to validate the homogeneity of the sample without air bubbles and to get the coordinates as a starting point for a follow-up laser ablation process to create a cylinder for high-resolution measurement by Nano-CT and FIB-SEM.(e) 3D rendering of reconstructed medium resolution Micro-CT. Note that the indicated scale bars are referring to the same measure of 100 µm.

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tion enabling a subsequent detailed phase analysis, which₃₈₅
was not possible with gray-value based analysis methods₃₈₆
only.

351 2.3. Image Processing Strategies

As already mentioned in Sec. 2.2, the described sample³⁹⁰ 352 preparation workflow is strongly supported by a detailed³⁹¹ 353 image processing strategy to allow for a robust identifica-³⁹² 354 tion and segmentation of particles consisting of different³⁹³ 355 constituent materials. In contrast to the workflow pre-³⁹⁴ 356 sented in Ditscherlein et al.[32], where single phase in-³⁹⁵ 357 formation is used to create particle-discrete data sets, in³⁹⁶ 358 the current study, the particles consist of different min-³⁹⁷ 359 eral phases, i.e., grains of different chemical composition.³⁹⁸ 360 Thus, the novelty of the present approach is the combina-³⁹⁹ 361 tion of a phase-discrete and an enhanced particle-discrete 400 362 workflow. The results from the phase-discrete and the⁴⁰¹ 363 particle-discrete analysis are based on different image pro-402 364 cessing workflows and, therefore, can be clearly separated⁴⁰³ 365 from each other as described in the next two sections. The⁴⁰⁴ 366 results of both approaches are merged in Sec. 3, to describe 367 the data sets in a meaningful way with regard to their dis-405 368 406 tributed descriptors. 369

370 2.3.1. Nano-CT Phase Segmentation

The BSE and EDXS analysis from the FIB slices was^{409} 371 utilized to add information to the Nano-CT data set seg- $^{\scriptscriptstyle 410}$ 372 mentation, according to the routine described by Lenz et⁴¹¹ 373 al. [33]. This was conducted in the commercial software⁴¹² 374 Arivis Vision4D with a machine learning algorithm based⁴¹³ 375 on ilastik [34]. The EDXS analysis was used as ground⁴¹⁴ 376 truth. Large phases, which could accurately be allocated⁴¹⁵ 377 in the EDXS maps, were identified in the corresponding⁴¹⁶ 378 Nano-CT slices and then manually segmented to train the⁴¹⁷ 379 machine learning algorithm, as illustrated in Fig. 7(c).⁴¹⁸ 380 To avoid an overlapping of regions of darker contrast in⁴¹⁹ 381 the Nano-CT reconstruction, whose contrast is affected by $^{\scriptscriptstyle 420}$ 382 edge gradients of brighter particles due to the optical reso-421 383 lution limit of about 150 nm, the separation threshold was⁴²² 384

set in such a way that it is able to separate the particles from each other. Manual color segmentation and, in particular, the threshold were adjusted until the resulting segmentation matched the corresponding EDXS analysis (cf. Figs. 7(d-e)). The difference between EDXS and segmented data was judged by visual inspection. Additionally, the Intersection over Union (IoU) [35] was determined on single-phase particles of adjacent slices as test data to the original training data to obtain a quantitative measure for the precision of the phase segmentation. The phases were color segmented manually in the slices of the test data and compared with the segmented data from the training. The IoU was calculated using the ratio of the overlap to the union of both data using $IOU = \frac{area_{overlap}}{area_{union}}$. In the manuscript, the segmented phases contained in each particle are referred to as grains of the particle and are counted individually for the later analysis, see Fig. 10. So, a single particle can contain multiple grains of the same phase. The individual particles are characterized by the image processing procedure described in the following section.

2.3.2. Nano-CT Particle Segmentation

In order to subsequently analyze the characteristics of particles, an image segmentation procedure was deployed which partitions the Nano-CT image data into regions corresponding to individual particles. The segmentation workflow is illustrated in Fig. 3. For that purpose, a fully convolutional neural network architecture was chosen, specifically an approach based on a 3D U-net (see the illustration in Fig. S5) is used, cf. [36, 37]. For the training of this network, three separate Nano-CT slices were annotated manually (each with around 100 individual particles) and used as ground truth. After training, the network is applied to the Nano-CT volume, and after binarizing the output of the trained network, individual particles are separated and can be identified by finding the connected components, see Fig. S6. As can be seen in Fig. 3, the manual annotation of the training data introduces inaccuracies that impact the network output, especially with respect to particle size and shape in areas where particles interface.470
Therefore, to capture the particle size and shape more ac-471
curately, a marker-based watershed algorithm is applied
to the binarized CT image data, using the connected components of the initial segmentation as markers, see [38].⁴⁷²
Further details on the network architecture and training₄₇₃

 $_{429}$ procedure can be found in [37].

430 2.3.3. Micro-CT Particle Segmentation

476 The phase and particle characterization emits in this 431 study mainly from the Nano-CT reconstructions. To esti-432 . 478 mate the local representativeness of the measured Nano-433 479 CT volumes with respect to homogeneity, the Micro-CT 434 480 reconstruction was utilized to bridge the scale of the parti-435 cle characterization towards larger dimensions. Here, the 436 482 Micro-CT volume was segmented by the machine learn-437 ing algorithm based on ilastik [34] similar to the Nano- CT^{483} 438 phase segmentation routine, but targeting only the parti-439 cle size without phase information. 440 486

441 2.3.4. Mineral Liberation

488 As mentioned in the previous sections, a particle $can_{_{499}}$ 442 consist of several connected components of a given phase, $_{_{490}}$ 443 referred to as a grains. Multiple grains of a particle phase $_{491}$ 444 are individually evaluated provided that they are sepa-492 445 rated from each other by another phase. The volume $\operatorname{frac-}_{_{493}}$ 446 tion (being equal to mass fraction due to similar densities)₄₉₄ 447 of a valuable mineral, denoted by i, that is present in a_{495} 448 collective of grains in an ungrown form, i.e., in the form of $_{_{496}}$ 449 free grains, is called the liberation degree and denoted by 450 LD_i . It is given by 451 497

$$\mathrm{LD}_{i} = \frac{V_{i,\mathrm{free}}}{V_{i,\mathrm{free}} + V_{i,\mathrm{intergrown}}}, \qquad (1)_{_{499}}^{^{498}}$$

500 where $V_{i,\text{free}}$ denotes the total volume of the fully liberated 452 particles (with volume fraction larger than 0.99) of phase $\frac{500}{502}$ 453 *i*, i.e., particles consisting of only one grain of the phase 503454 *i*, and $V_{i,\text{intergrown}}$ denotes the volume of the remainder $_{504}$ 455 of phase *i*. However, smaller particles are generally more $_{505}$ 456 likely to occur fully liberated. Thus, investigating size 457 dependent liberation information is of interest. Therefore, $\frac{1}{507}$ 458 for an interval $I \subset [0, \infty)$, i.e., a defined particle size range, $\int_{508}^{508} f_{508}$ 459 the liberation degree of particles with volume in I is given 509 460 by 461 510

$$\mathrm{LD}_{i}^{I} = \frac{V_{i,\mathrm{free}}^{I}}{V_{i,\mathrm{free}}^{I} + V_{i,\mathrm{intergrown}}^{I}},$$
(2)512

where $V_{i,\text{free}}^{I}$ denotes the volume of the fully liberated par-⁵¹⁴ ticles with volume in I of phase i and $V_{i,\text{intergrown}}^{I}$ denotes ⁵¹⁵ the volume of the remainder of phase i in particles with ⁵¹⁶ volume in I. Thus, partitioning the size range into dis-⁵¹⁷ joint intervals, the liberation degree can be analyzed in ⁵¹⁸ more detail, see Fig. 10.

Usually, the shares are determined by creating property classes, e.g. via swim-sink sorting or via 2D image analysis. However, in the letter case only area fractions are being considered instead of volume fractions.

3. Results

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3.1. Correlative Microscopy Workflow

The first challenge is the identification and correlation of the minerals (particles and grains) across the different congruent reconstructed volumes or acquired slices using different tomography and imaging techniques. As can be seen in Fig. 4, the applied workflow starts with the medium-resolution Micro-CT scan of a sample with larger size of around 400 µm. In a next step, the Micro-CT pillar is site-specifically cut down using laser ablation to fit the Nano-CT's field of view of $(65 \,\mu\text{m})^2$ enabling a correlation with the Nano-CT reconstruction. Then, three vertically stitched high-resolution Nano-CT tilt series covering the complete Nano-CT pillar are acquired. Their subsequent 3D reconstruction is further identified and correlated with the corresponding region in the Micro-CT tomogram. In this way, a certain volume within the reconstruction obtained by Micro-CT can be displayed at higher resolution and with improved contrast. In the last preparation step, the Nano-CT pillar is FIB milled from the vertical side creating a cut parallel to the pillar axis. This enables SEM-EDXS analyse of a specific region of interest (ROI) within the pillar, so that for this ROI, measurement data from all three microscopy techniques is available and can directly be correlated, as illustrated in Fig. 4.

3.2. Precise Mineral Allocation

The distinction between minerals in the Nano-CT reconstructions can be estimated to certain extent already by the local grav values of voxels (cf. Figure 4(a-b)), since each component exhibits a specific attenuation coefficient for the 5.4 keV X-ray beam, in contrast to the Micro-CT instrument, see Tab. 1. Although the X-ray beam is quasimonochromatic, the corresponding gray values in the 3D reconstruction for the different phases are further affected by a remaining X-ray energy dispersion, material density variations or limited resolution. Thus, local attenuation coefficients are not perfectly recovered in the 3D reconstructions, which impedes a unambiguous assignment of a certain mineralogical phase to every voxel only from gray levels. To compensate this uncertainty, we utilize local compositional information from SEM-EDXS analyses to inform the Nano-CT volumes and extrapolate the phase information from a 2D slice to a 3D reconstructed volume. This method allows for a thorough assignment and segmentation of the local mineralogical phase information to every voxel of a 3D reconstruction and therewith enables a precise identification of distinct particles and grains.

The analysis workflow with the achievable sample size and resolution of each step and technique is illustrated in Fig. 5. The sample is site-specifically sliced using the FIB with nearly horizontal cuts (slices perpendicular to



Figure 3: Exemplary particle segmentation workflow using U-nets. A grayscale image is used as input. Output, ground truth and weight map lead to a loss function. Afterwards the parameters are updated and the cycle repeats with a new image as input.



Figure 4: Correlative microscopy workflow of one identical region across multiple length scales. Virtual slices through (a) medium-resolution Micro-CT and (b) high-resolution Nano-CT reconstruction revealing the same particles as detected in (c) the BSE image after the smaller Nano-CT pillar (diameter of 60 µm) was vertically cut out of the larger Micro-CT sample. The insets illustrate the increased detail depth of the BSE images compared to the Micro-CT and Nano-CT slices. The EDXS maps (net intensities) provide additional information about the chemical composition (saxolite in red, dolomite in yellow, magnesite in green, talcum in cyan; see also Fig. 6).



Figure 5: Correlative workflow from Micro-CT measurements of a larger volume to a sample with reduced size generated by applying laser ablation preparation suitable for Nano-CT measurements, both realistically reconstructing the 3D morphology of the particulate sample. The application of an additional milling process with FIB enables the acquisition of slices through the volumes used for Nano-CT for chemical characterization via SEM-EDXS.

the long axis of the pillar sample) and analyzed with SEM-555 523 EDXS (cf. Fig. 5 center) after the correlation of the Nano-556 524 CT volumes with the SEM image of a vertical slice through 557 525 the Nano-CT pillar (cf. Fig. 4(b)). Fig. 6 depicts the558 526 SEM-EDXS analysis of an exemplary, nearly horizontal⁵⁵⁹ 527 slice. The cutting heights are chosen according to inter-560 528 esting slices identified in the Nano-CT tomograms, where 561 529 every possible constituent mineral appears to a sufficient 562 530 amount enabling a reliable segmentation of all phases. In₅₆₃ 531 Fig. 4 (vertical slice in (b) and (c)) and Fig. 7 (horizontal⁵⁶⁴ 532 slices in (d) and (f)), the same particle shape in both im-565 533 ages proves the correctly applied correlation. The phase-566 534 discrete segmentation is processed on the horizontal slices, 567 535 shown in Fig. 7 where every cut was taken from one of₅₆₈ 536 the three individual Nano-CT volumes. Each mineral re-569 537 sults in a different gray-scale value (cf. Fig. 7, (c) and (f))570 538 and can be directly matched with a combination of the571 539 EDXS signals leading to the color code used in Figs. 6 and 572 540 7, where red represents saxolite (calcite), yellow dolomite, 573 541 green magnesite and cyan talcum. Fig. 6 shows (a) a BSE₅₇₄ 542 contrast image and element-specific EDXS maps of (c) cal-575 543 cium, (d) magnesium and (e) silicon with respect to the576 544 constituent minerals of one exemplary cut. This leads to577 545 the combined EDXS image in Fig. 6(b) with the same color578 546 code enabling a clear/convenient distinction between the 579 547 different constituent minerals. 580 548

549 3.3. Volume Correlation

The element-specific EDXS signals of calcium, mag-⁵⁸³ nesium and silicon are analyzed with respect to the con-⁵⁸⁴ stituent minerals, as illustrated in Fig. 7(a). First, the⁵⁸⁵ machine learning (ML) segmentation algorithm (see Sec.⁵⁸⁶ 2.3.1) is trained in a selected Nano-CT slice (Fig. 7(a)) to⁵⁸⁷

assign every voxel of the segmented particle regions to one of the four minerals. Here, the particles in the EDXS analvsis (as indicated with a blue rectangle), which are large enough to be visually allocated to the phases, are utilized for the training. The larger phases from the EDXS maps can clearly be identified and labeled in the corresponding Nano-CT slices (cf. Fig. 7(b)). Here, typically 1-3 large particles of each phase can be colored. However, the exact shape slightly differs due to surface cutting artifacts of the Ga-ion beam (beam divergence). Phases, like the vellow dolomite at the bottom of Fig. 7(c), typically also extend into z-direction (long axis of pillar) and therewith can be identified also in other nearby z-slices, around 2-5, of the Nano-CT tomogram. Likewise expands the EDXS analysis over adjacent z-slices of the Nano-CT tomogram. Thus, the manual labeling is repeated for multiple Nano-CT slices which contain the particle with known phase information from EDXS. The trained algorithm is then applied to the entire Nano-CT volume. The final results show the phase segmentation of each of the particles with respect to saxolite, dolomite, magnesite and talcum in Fig. 7(c and e). A closer look at the identified phases in the blue boxes of Figs. 7(a-c), shows that the actual composition of particles can be more complex than originally suspected. The larger red particle, for instance, had been originally identified in the manually segmented Nano-CT data set for machine learning training as one pure saxolite grain (cf. Fig. 7(b)). However, after the training with the additional SEM-EDXS data, the actual composition results in a mixture of red and vellow (saxolite and dolomite) grains, as shown in Fig. 7(c). In this way, it is possible to refine the ML algorithm until the qualitative difference by visual inspection between the ML segmentation results

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Figure 6: Mineral determination according to the chemical components. (a) BSE overview image already enables to distuinguish individual particles with different chemical composition. (c) Ca- (d) Mg- and (e) Si-EDXS analysis (net intensities) reveals the individual components in the combined image (b) with high precision. See Fig. S4(b) for the slice position indicated in the Nano-CT volume.



Figure 7: Phase-related segmentation workflow. (a) SEM-EDXS analysis (net intensities) reveals the distinct mineral phases through their different chemical composition (saxolite in red, dolomite in yellow, magnesite in green, talcum in cyan). (b) Virtual slice through the Nano-CT reconstruction at the same location as the corresponding EDXS map: gray scale range of each phase in Nano-CT is manually segmented to train the ML algorithm in individual slices using the EDXS signal. (c) The ML algorithm allocates the different phases first to one single slice and from this to the complete corresponding Nano-CT reconstruction. This is repeated at three other positions (one slice in each single Nano-CT reconstruction) shown in (d) and (f), leading to a phase-sensitive segmentation in (e) which is extrapolated to the full Nano-CT reconstruction, as illustrated in Fig. 8. The difference between the exact particle morphology of (d) and (f) is a result of the divergence of the focused ion beam. All scale bars represent 10 µm. See Fig. S4(b) for the slice positions and the positions of the single Nano-CT reconstructions indicated in the complete Nano-CT volume.



Figure 8: Three stitched Nano-CT reconstructions as gray-scale im-618 age in (a) and after phase-discrete segmentation in (b).

and the EDXS maps is minimized. To judge the refined $^{\rm 621}$ 588 training quantitatively, we color segmented single-phase $^{\rm 622}$ 589 particles of each mineral manually in an untrained z-slice⁶²³ 590 and calculated the IoU of the manual and ML segmenta-⁶²⁴ 591 tion results. The IoU ranges between 0.95 and 0.84, with⁶²⁵ 592 a mean value of 0.91 for the utilized particles. The final $^{\scriptscriptstyle 626}$ 593 segmentation of each of the three Nano-CT volumes is il^{-627} 594 lustrated in Fig. 7(d-f). The EDXS maps in Fig. 7(d) and $^{\rm 628}$ 595 the virtual slices in Fig. 7(f) show a good match and the 629 596 resulting segmentations in the three slices in Fig. 7(e) are $^{\rm 630}$ 597 in a good agreement with Fig. 7(d). Lastly, as illustrated $^{\scriptscriptstyle 631}$ 598 in Fig. 8(b) and in Fig. S4, the three individual segmented $^{\scriptscriptstyle 632}$ 599 sub-volumes are merged to obtain the final phase-discrete $^{\rm 633}$ 600 634 segmentation of the whole Nano-CT reconstruction. 601 635

602 3.4. Particle-discrete Analysis

After having finished the phase-discrete segmentation, we investigate the volumes with respect to particle-discrete information. To do so, the identified regions in the Nano-CT volumes of saxolite, dolomite, talcum and magnesite are registered in the particle-wise segmentation and the



Figure 9: Mass-weighted, normalized histograms of the equivalent spherical diameter of particles which predominantly consist of one mineral (volume fraction larger than 0.5) and of all identified particles. The lower end is cut off by the resolution.

resulting mineralogical composition of individual particles is correlated to particle descriptors, such as the equivalent spherical diameter (see also the comparison of phasediscrete and particle-discrete segmentation illustrated in Fig. S6). The mineralogical composition of each particle is quantified by four scalar values, corresponding to the volume fractions of saxolite, dolomite, talcum and magnesite of the given particle. Fig. 9 depicts the mass-weighted, normalized histograms of the equivalent spherical diameter of particles which predominantly consist of one mineral (volume fraction larger than 0.5). The combination of the segmentation workflows considered in Sections 2.3.1 and 2.3.2 provides a volume ratio of 27 to 73 (saxolite to talcum + dolomite + magnesite) which is comparable to the original mixture of minerals (see Sec. 2.1) and will be further discussed below.

While almost all particles consist predominantly of one mineral, this does not mean that they occur fully liberated. The latter can be quantified by the liberation degree of a mineral, which is the ratio of the volume of this mineral that is liberated to the total volume of this mineral, as introduced in Eqs. (1) and (2). The liberation degree varies depending on particle size, as shown in Fig. 10, where smaller particles generally have a higher probability of occurring fully liberated. However, the degree of liberation only provides a simplified view on the collected data, which is also attainable by the application of standard methods, e.g., multi-stage density separation by heavy liquids or thin sectioning in the case of a mineral liberation analyser. But, since our method provides comprehensive particle-discrete data for all 22468 identified particles in the Nano-CT reconstructions, we can also plot particle and grain volume against each other, see Fig. 11, where each saxolite grain is considered individually in order to predict a possible sepa-

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Figure 10: Volume specific liberation degree. The values at the dots are computed by means of Eq. (2) for the surrounding particle size interval.

ration process. Here, the corresponding volumes illustrate 642 a more apprehensible relation than the sphere equivalent 643 diameter and give hint directly to the total amount of par-644 ticles that are not fully liberated. The results obtained for 645 the other three minerals are illustrated in Fig. S7. The red 646 curve in Fig. 11 provides an estimation of the liberation 647 degree by its distance to the diagonal line – the farer away 648 from the diagonal line, the lower the liberation degree. 649 The corresponding statistical error (gray shadowing) gives 650 hint to the reliability for each fixed particle size: the error 651 is small in the range from $2\,\mu\text{m}^3$ to $200\,\mu\text{m}^3$ and increases 652 for smaller and larger particles because of the resolution 653 and FOV limits, respectively. 654

655 3.5. Scale-bridging Tomography

The analysis of the Nano-CT data results in detailed 656 information with respect to size and mineralogical compo-657 sition of the segmented particles, as depicted in Figs. 9 and 658 11. However, in case of a material with inhomogenously 659 distributed particles, results derived from the analysis of 660 Nano-CT-sized volumes might not be representative. To 661 investigate the homogeneity of the material, the Micro-CT 662 volume is analyzed, which delivers an approximately 60-663 times larger acquired sample volume compared to the three 664 combined Nano-CT measurements. For that purpose, the 665 particle-discrete segmentation of the Micro-CT volume, as 666 described in Sec. 2.3.3, is decomposed into 60 pieces of 667 the same size as a Nano-CT reconstruction. Then, distri-668 butional properties of particle sizes within each of these 669 pieces are investigated for given size intervals, see Figure 670 12. In particular, for each Nano-CT-sized volume piece, 671 the frequency of particles whose equivalent spherical di-672 ameter belongs to a certain size interval is computed. The 673 mean values over all volume pieces and the corresponding 674 95% confidence intervals are depicted in Fig. 12. 675



Figure 11: Grain over particle volume map targeting saxolite, which corroborates the segmentation workflow being able to analyze individual particles. The red curve illustrates the mean grain volume for a given particle volume, whereas the gray shadowing indicates the corresponding error bar.



Figure 12: Distributional properties of the equivalent spherical diameter of particles in Nano-CT-sized pieces of the Micro-CT tomogram. In particular, the mean values of frequencies over all pieces are depicted as blue horizontal lines for each size interval under consideration. The corresponding standard errors are depicted by purple vertical lines and the 95 % confidence intervals are visualized in gray. The yellow vertical line represents the cut-off particle size resulting from the medium resolution of the Micro-CT scan of about $3.2 \,\mu$ m (each size interval comprises the diameter range surrounding each sampling point in the graph from the half-distance towards the preceding sampling point to the half-distance towards succeeding sampling point, respectively).

This visualization displays the variability of the esti-730 676 mated particle size distribution when considering Nano-731 677 CT-sized volumes—allowing for the assessment of the re-732 678 liability of Nano-CT-based characterizations. Since the733 679 95% confidence intervals are small for all considered size₇₃₄ 680 intervals, we can conclude that all Nano-CT-sized volume735 681 pieces are similar with respect to the constituent particle736 682 sizes and number of particles without comparing the min-737 683 eral composition, so that one single investigated Nano-CT₇₃₈ 684 volume can be considered as representative for the origi-739 685 nal Micro-CT sized sample. For further extrapolation to740 686 a bulk sample, we refer to a representative study from the741 687 Micro-CT perspective on the applied preparation workflow₇₄₂ 688 [19], showing the homogeneity for equally prepared sam-743 689 ples. The lower resolution of the Micro-CT reconstruction,744 690 in contrast to the Nano-CT measurements, partly leads745 691 to wrongly-interpreted, connected particles and disregards746 692 smaller particles (below 3.2 µm). Thus, the determined to-747 693 tal number of particles in the Nano-CT-sized volume pieces748 694 of the Micro-CT reconstruction is smaller than in corre-749 695 sponding Nano-CT measurements of the same volume size.750 696 Similarly, the analysis of Micro-CT reconstructions leads751 697 to a right-shifted particle size distribution. 698 752

699 4. Discussion

The presented workflow enables a quantitative charac-⁷⁵⁶ 700 terization of the considered particle system with respect to⁷⁵⁷ 701 particle size and composition. Note that for the compo-758 702 sitional characterization of particles, FIB slices and SEM-759 703 EDXS analyses are required to obtain a phase segmenta-⁷⁶⁰ 704 tion of the considered volumes, see Sec. 2.3.1. Due to the⁷⁶¹ 705 divergence of the FIB beam and the subsequent curvature⁷⁶² 706 of the cutting plane, it is difficult to precisely associate pix-⁷⁶³ 707 els in the EDXS data with corresponding voxels in Nano-⁷⁶⁴ 708 CT reconstructions. Therefore, we did not directly utilize⁷⁶⁵ 709 the EDXS data itself for training the considered ML algo- 766 710 rithm. Instead, large particles within the EDXS data have⁷⁶⁷ 711 been manually identified and labeled within the Nano-CT⁷⁶⁸ 712 reconstructions—which provided us with training data for⁷⁶⁹ 713 the segmentation procedure. 770 714

In principle, it would be possible to slice and view⁷⁷¹ 715 (including SEM-EDXS) the whole sample volume by FIB⁷⁷² 716 milling (so-called slice & view FIB-SEM tomography). How²³ 717 ever, the acquisition of FIB-SEM tomography data for⁷⁷⁴ 718 such large sample volumes with a high spatial resolution⁷⁷⁵ 719 is a tedious and time-consuming procedure which addi-776 720 tionally often suffers under charging effects. Our workflow⁷⁷⁷ 721 demonstrates a more time-effective, alternative investiga-778 722 tion routine using EDXS maps from only a few selected⁷⁷⁹ 723 slices to inform the complete Nano-CT volume segmenta-780 724 781 tion. 725

The limitations of the presented routine with respect⁷⁸² to the Nano-CT experiments are the spatial resolution and⁷⁸³ the gray value range in the reconstructed volumes caused⁷⁸⁴ by the quasi-monochromatic beam (including noise). The⁷⁸⁵ ⁷⁸⁶ lower limit of the particle size depends on the optical resolution of the Nano-CT reconstructions (about 150 nm in the used imaging mode). Due to this reason, particles with an equivalent spherical diameter smaller than two times the resolution, so $<0.3 \,\mu\text{m}$, were omitted in the performed analyses. The exact monochromatic filtering of the Nano-CT microscope is complex. Therefore, we provide an experimental approximation for the limitation distinguishing different mineral components according to their attenuation length difference. In this case study, the difference in the attenuation coefficients of the four minerals is large enough to be able to separate the phases well (cf. Tab. 1). For instance, the difference in the reconstructed gray values between saxolite and dolomite is 4000 corresponding to 8 µm attenuation length (see Tab. 1), which is around ten times larger than the gray value range of 400 within the saxolite phase. Similar to the particle size limit, we would suggest a minimum difference in the attenuation length resulting in a gray value difference of two times the gray value range of the individual phases, which is 800 in the case of saxolite. This value range approximately equals an attenuation length difference of $\pm 1.6 \,\mu\text{m}$, meaning that all phases with a higher difference in grav value or attenuation length should be distinguishable in the case of this Nano-CT instrument and the applied imaging conditions.

Note that the applied Nano-CT instrument is not able to image the whole 400 µm sample due to its limited field of view and the absorption length of the employed X-ray energy. Therefore, if the sample was inhomogeneous, the analysis of Nano-CT data could lead to unrepresentative results. In order to verify homogeneity of the sample, the available Micro-CT image data was analyzed regarding the representability of Nano-CT sized volumes. More precisely, Nano-CT-sized volumes taken from the Micro-CT volume indicate similar particle size distributions, see Fig. 12. Therefore, we assume that the position of the Nano-CT measurements in the considered sample has a marginal impact on the results. This is further corroborated by observing a similar trend of the particle size distributions above 3 µm equivalent spherical diameter for the investigated Micro-CT and Nano-CT reconstructions (cf. Figs. 9 and 12). However, we expect an overall shift towards larger particle sizes due to the reduced resolution in the Micro-CT volume.

As described in Sec. 2.1, the mixture was intentionally prepared with the volume ratio of 30 to 70 (talcum to saxolite). Our workflow enables us to determine the volume ratio from reconstructed 3D image data. More precisely, since we can assign for each voxel in the Nano-CT data the corresponding mineralogical phase, we can compute the phase volume fractions, which leads to a volume ratio of 73 to 27 (talcum to saxolite). This is in good agreement with an inadvertently prepared ratio of 70 to 30, and not as expected 30 to 70. Although a precise quantification of the acquired EDXS maps from single slices (cf. Figs. 6(b) and 7(a and d)) w.r.t. the volumetric presence of the distinct minerals is difficult to pursue, e.g., due to

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the inclined sample geometry and the detector position⁸⁴² 787 (absorption correction), the EDXS maps rather indicate as43 788 ratio of 70:30. This is further supported by the determined⁸⁴⁴ 789 area ratio of the segmented saxolite regions in the EDXS⁸⁴⁵ 790 maps w.r.t. the talcum phases (average area ratio of 65:35846 791 for three EDXS maps in Fig. 7(d)). Assuming a sample⁸⁴⁷ 792 was prepared as 70 to 30 and considering that, according⁸⁴⁸ 793 to the manufacturer, the original saxolite powder also con-849 794 tains talcum (4%), the actual volume ratio of the mixture⁸⁵⁰ 795 should be 71.2 and 28.8 which matches the estimated re-851 796 sult obtained from image data even better, demonstrating852 797 the precision of the presented workflow. 853 798

The particle-wise segmentation of the Nano-CT recon-854 799 structions is based on a convolutional neural network, and⁸⁵⁵ 800 thus, the network's output is unpredictable for images⁸⁵⁶ 801 that exhibit novel features not seen before in the training857 802 data. For the current data set, it was possible to encom-858 803 pass relevant features by labeling just three slices, but this859 804 might not be possible for image data with more diversesso 805 images and features. In addition, while errors caused by₈₆₁ 806 over-segmentation are negligible within our particle-wises 807 segmentation (observed through visual inspection), under-863 808 segmentation can occur especially when small particles are⁸⁶⁴ 809 involved. Since we consider volume-weighted descriptors,865 810 this type of under-segmentation should not significantly₈₆₆ 811 influence the results. However, the occurrence of compos-867 812 ite particles is overestimated as under-segmentation might*** 813 occur for neighboring particles composed of different min-869 814 erals. To account for this, particles which have a volumes⁷⁰ 815 fraction greater than 0.99 for any mineral were considered⁸⁷¹ 816 to be fully liberated. 872 817

In comparison to common chemical analyses, the ad-818 ditional particle-discrete information extracted from our 819 correlative workflow offers enhanced knowledge on the in-820 vestigated materials. For example, note that in mineral₈₇₄ 821 processing similar particle systems occur as a result of par-822 ticle milling processes which are subsequent to the mineral $_{\rm s76}$ 823 excavation to increase the LD of the material. Thus, on₈₇₇ 824 the basis of the derived particle-size related data, it is pos-825 sible to adjust the applied particle milling process if the 826 required LD of the minerals is reached at a certain par-827 ticle size. For that purpose, the degree of liberation was⁸⁷⁹ 828 analyzed for different minerals and particle sizes. In par-829 ticular, for saxolite the degree of liberation decreases for 830 particles larger than $0.5\,\mu\mathrm{m}$, as shown in Fig. 10. Such⁸⁸¹ 831 a trend is expected due to the higher stability of $\mathrm{single}^{^{882}}$ 832 grained particles and it can guide towards an optimized⁸⁸³ 833 choice of the milling size. More specifically, if a high de-834 gree of liberation is desired for saxolite, further milling⁸⁸⁵ 835 might be necessary to increase the LD. 836 887

837 5. Conclusions

In summary, the presented correlative characterization workflow over multiple length scales is utilized to unanbiguously allocate and identify different constituent mineral types contained as grains in similarly sized particles. This can be a challenging task for common mineralogical characterization methods which rely on only particle sizes. In particular, a large volume was reconstructed and analyzed in 3D with various imaging techniques, in order to ascertain a comprehensive characterization of the particles' size and composition. Our presented correlative 3D characterization approach is suitable to be applied to particulate samples with primary particle sizes in the range of $0.3\,\mu\text{m}$ to $15\,\mu\text{m}$ by considering resolution and FOV limits of the Nano-CT instrument. In our study, the morphology and composition are available for each individual particle and can be evaluated to directly compute the particle-wise mineral liberation, which usually cannot be determined using standard methods for particle characterization. The combination of multiple imaging techniques significantly improves the statistical relevance (high number of particles) and segmentation precision (high resolution) compared to the application of a single technique. The proposed method provides multidimensional particle properties enabling a more detailed understanding of multidimensional separation processes, see [39]. In a forthcoming study, the presented characterization workflow is supposed to be deployed to evaluate the separation efficiency achieved by a combination of deflector wheel classifier and triboelectric charge sorting. Furthermore, the determined mineral liberation degree properties will allow tuning processes to obtain optimized milling degrees of the original mineral batches. Moreover, the data obtained with the presented approach can be used as input for numerical simulations to obtain a detailed understanding of these complex separation processes.

Data Availability

The reconstructed image data, in this case, TIFF stacks, the related acquisition and reconstruction parameters are stored within the scientific data repository of Technische Universität Dresden and TU Bergakademie Freiberg with all relevant meta-data [40].

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screening and in situ mechanical testing of structural and⁹⁶⁰
functional materials" (Project-ID 316992193), the Collab-⁹⁶¹
orative Research Centres 1411 "Design of Particulate Prod-⁹⁶²
ucts" (Project-ID 416229255) and 1452 "Catalysis at Liq-⁹⁶⁴
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